

7-15-2011

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Aftab Alam

University of Illinois at Urbana-Champaign

Suffian N. Khan

Ames Laboratory

Brian G. Wilson

Lawrence Livermore National Laboratory

Duane D. Johnson

Ames Laboratory, ddj@iastate.edu

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Disciplines

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Comments

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Efficient isoparametric integration over arbitrary space-filling Voronoi polyhedra for electronic structure calculations

Aftab Alam,^{1,3} S. N. Khan,^{2,3} Brian G. Wilson,⁴ and D. D. Johnson^{1,2,3,*}

¹*Department of Materials Science and Engineering, University of Illinois, Urbana, Illinois 61801, USA*

²*Division of Materials Science and Engineering, Ames Laboratory, Ames, Iowa 50011, USA*

³*Department of Physics, University of Illinois, Urbana, Illinois 61801, USA*

⁴*Lawrence Livermore National Laboratory, Livermore, California 94550, USA*

(Received 1 March 2011; revised manuscript received 9 May 2011; published 6 July 2011)

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DOI: [10.1103/PhysRevB.84.045105](https://doi.org/10.1103/PhysRevB.84.045105)

PACS number(s): 02.60.Jh, 71.15.Dx

I. INTRODUCTION

A variety of science and engineering research problems require multicentered integrals that cannot be solved analytically due to the complex domains of integration. The total energy and potential in any site-centered, electronic-structure calculation involves the evaluation of three-dimensional integrals over convex Voronoi polyhedra (VP).¹ Although a number of numerical integration techniques have been proposed,²⁻⁶ an efficient, accurate, reliable, and easily implemented scheme is still lacking. Prior methods often rely on detailed analysis of symmetry properties of the integration domains and, hence, limit their applicability to arbitrary atomic geometries and structures. A major continuing need is an integration method over space-filling VP that has a high degree of accuracy with a minimal computational effort and that is sufficiently generic so that it can be used in most electronic-structure application codes.

For example, the “exact” linear muffin-tin orbital (EMTO) method⁷ uses an approach from Gonis *et al.*⁸ to overcome VP integration issues for the Poisson potential, but it is extremely slowly convergent; various Korringa-Kohn-Rostoker- (KKR) based codes, such as the linear-scaling multiple-scattering (LSMS),⁹ utilize shape functions to perform VP integrations, which, as we show, is slowly convergent and limited in accuracy; the full-potential linear augmented wave¹⁰ (FLAPW) method avoids VP integrals (via nonoverlapping muffin-tins and Fourier methods over the entire unit cell), but never determine site-VP-specific properties and require a larger number of spherical harmonic basis functions and a huge number of plane waves.

We present such an algorithm by combining a weighted VP tessellation^{1,11} with isoparametric methods¹² to provide rapidly convergent integrals for various integrands, including Coulomb singularities. For generality, we use a radical plane construction¹³ (RPC) or power diagrams¹⁴ to guarantee convex, space-filling VP. For electronic-structure use,

physics-based weights are optimally chosen as ratios of radii determined from the topology of the electronic density.¹⁵ Isoparametric transformations then permits analytical mapping of polyhedra subdomains to a bi-unit cube, which are then simply integrated by Gauss-Legendre quadratures. For any VP we only need evaluate numerically the integrands, Jacobian, and weights at Gauss points for a relatively fast and accurate integral, and with no issues regarding divergence.

We showcase our isoparametric method by two means. First, we evaluate various electronic integrals analytically using a well-known charge-density model by van Morgan,¹⁶ and show directly the accuracy and efficacy of our numerical method. Second, we implement the method in an all-electron KKR code¹⁷ and apply it to a phase stability study of face-centered-cubic, (dis)ordered FePd. We exemplify the accuracy for formation enthalpies and the insensitivity of results to the choice of spherical-harmonic basis set (L expansion) due to the use of weighted VP.

We organize the paper as follows: After Sec. II on the background, we describe in Sec. III the RPC tessellation and weights that we merge with an isoparametric integration via an analytic dual-coordinate transformation, known in the finite-element community, to create a general and optimal integration scheme. In Sec. IV, we describe the van Morgan charge-density model to assess the performance of any integration method. In Sec. V, we address the accuracy, convergence, and timings of this isoparametric scheme for close-packed structures; machine-precision accuracy with expected convergence is found, with millisecond timing for each VP. Our approach is greater than 10^5 times faster and 10^7 times more accurate than that with shape functions. Finally, we discussed the results for application to FePd, then conclude in Sec. VI.

II. BACKGROUND

Given any atomic configuration, the first step to compute any site-centered integral quantity is to perform a Voronoi

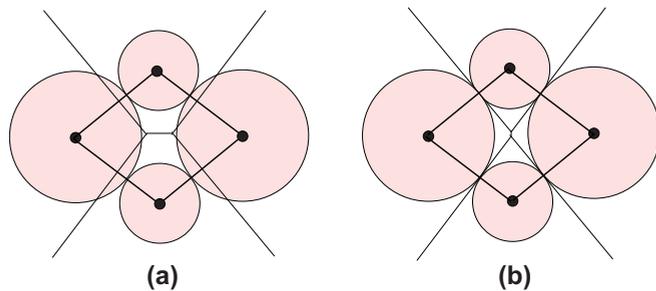


FIG. 1. (Color online) Tessellation via (a) geometry and (b) RPC. The “size” of little (big) atoms is over (under) estimated in (a), affecting properties, whereas (b) reflects atomic “size” if radii are determined from charge densities (see text).

tessellation of the space in and around the molecule or solid, including possible empty sites to improve, for example, a site-centered basis set. Standard geometric (i.e., the Wigner-Seitz) tessellation subdivides space into VP such that every point within a VP cell has the property of being closest to one and the same site, and that each polyhedral face is orthogonal to and bisects the line segments joining the site centers (see Fig. 1). However, in most materials (e.g., size-mismatched alloys), it is a poor subdivision of space, making VP that correspond to the smaller (larger) atoms too large (small);¹⁵ see Fig. 1. Such errors impact solid-state and biological problems where, for example, an accurate interstitial volume is required for reliable predictions of thermostability of cavity-filling mutants in proteins,¹⁸ or for statistical models in continuum systems where packing geometry plays a key role.¹⁹

Given these deficiencies, various proposals have been made to place the dividing plane subject to atomic size. Richard²⁰ suggested using the ratio of the distance between atoms and dividing plane to be equal to the ratio of the corresponding atomic radii, but this does not reflect bonding. RPC weights the distance to each atom by subtracting the squared atomic radius from the squared Euclidean distance,^{13,14} which guarantees convex and space-filling VP, but radii must be provided. Other generalizations include the introduction of nonplanar boundaries between atoms.²¹ For site-centered methods, the convex, space-filling property is critical; for example, in KKR the scattering matrices are only defined for convex VP, but the spherical-harmonic basis must reflect accurately the spherical density of each site or else the basis must be augmented (e.g., plane waves in FLAPW¹⁰ or spherical waves²²). Notably, the VP tessellation benefits from a judicious choice of weights related to electron density.¹⁵

With a Voronoi tessellation in hand, a scheme must be chosen to perform VP integrations. In one dimension, there are many numerical techniques that easily achieve accurate results. Yet, accurate and fast techniques for three-dimensional integrands with unusual domains such as those found in molecular or solid-state calculations remain an active area of study.²³ We now mention some key previous work: Ellis and Painter² used Diophantine integration²⁴ in molecular calculations, with convergence as $O(N^{-1/2})$ to $O(N^{-1})$ for N sample points—just better than Monte Carlo. Becke²⁵ used standard Voronoi partitioning with simplifications introduced to reduce multicenter integrals to a sum of single-center ones,

with less-detailed convergence studies than we provide here; we find that it uses slightly more points than ours for low ($\sim 10^{-3}$) accuracy and much slower convergence for higher precision. More recently, Gaussian product formulas have been found useful when awkward domains of integration were split into tractable subdomains.²⁶ It is, however, not only the form of the integration domain but also behavior of the integrand that may necessitate the use of product formulas and further subdivision of the subdomains.

For site-centered basis sets, there are limited choices of subdivision for convex VP domains. Baerends and coworkers^{3,5} broke each VP into subpolyhedra formed by the nucleus and its base of one of the VP faces followed by further subdivision of the face into connected sets of triangles and quadrilaterals. Averill and Painter⁴ cropped each VP by an inscribed sphere to form an interstitial region associated with each VP face; hence, interstitial integrals are expressed as a sum of integrals over cropped pyramids. (In finite systems, a separate subdomain is the bounding part of the space outside the local atomic VP.) We use an analytic transformation to a bi-unit cube for the cropped integrals; an approach similar, but not the same as, Baerends and coworkers,^{3,5} as we shall discuss. Also, unlike previous methods we use the underlying charge density to choose an optimal subdomain of integration. In all-electron methods, the wave function and potential have cusps and singularities near the nuclei, respectively; these functions are easily integrated over spherical domains, although the “interstitial” region (between the sphere and VP facet) has a complex shape. We utilize the charge-density topology and the behavior of the integrands to determine the VP (spherical and interstitial) subdomains.

Although various integration methods have been proposed, the accuracy attained has usually been poor compared to the computational effort expended, typically a modest (7–8 digit) accuracy required a large number of sampling points. The desire, of course, is to approach machine-precision accuracy with a modest number of sampling points. While noting some similarities with previous methods,^{4,5} the present approach is unique in features and, particularly, is efficient and accurate for polyatomic systems; also it has the advantage of being conceptually simple and easy to implement. We verify that accurate (14 digit) integration over various kernels is achieved with a modest number of Gauss points. Moreover, when combined with the use of physics-based weighted VP, an insensitivity to site-centered basis sets is possible while achieving high accuracy, as we show.

III. METHOD

The partitioning of space in and around a molecule or solid into convex polyhedra by RPC is described, followed by an analytic dual-coordinate transformation (isoparametric mapping) of a bi-unit cube to obtain the shape of any specific facet subdomain of each VP. Dissection of each VP can be accomplished in two ways: Either each VP (1) is divided into subdomains formed by the pyramid between face and a site’s origin; or (2) is split into an inscribed sphere domain and a sum of interstitial domains between the sphere and each facet plane. We then find the integral at each VP as a sum over the Gauss quadratures by numerical evaluation of the

integrands, weights, and Jacobian of the transformation. If the integrand is evaluated at known Gauss points, only sampling points determine the error; however, if a numerical (discrete) grid is used, then interpolation error needs to be ameliorated.

A. Weighted Voronoi radical plane construction

In unweighted tessellations, the subspace of points closer to an atom centered at \mathbf{r}_i than any other atom centered at \mathbf{r}_j is considered the VP about atom \mathbf{r}_i . For an atom on the edge of a molecule, the “polyhedron” will be unbounded; that can be amended by introducing an extra dividing plane that is tangent to the inscribed sphere of the unbounded cell. The best choice will be the plane that also minimizes the volume in the resulting cell. Of course, no such edge difficulties can occur for periodic crystals. In the case of a one-atom crystal, the VP is simply the Wigner-Seitz (geometric) cell.

For a crystal with atoms of unlike size, the standard Voronoi partitioning of space assigns too much volume to the smaller atoms, see Fig. 1(a), and the corresponding VP is unphysical.¹⁵ The biased partitioning incorrectly determines the atomic volumes and the VP structure. A weighted tessellation corrects by re-weighting the distances to the site centers. This resizing allows the volume given to each site to grow or shrink in accord with its size (radius). RPC uses the weighted metric $\|\mathbf{r} - \mathbf{r}_i\|^2 - R_i^2$, where R_i is the radius of the i th atom and $\|\mathbf{r} - \mathbf{r}_i\|$ is the Euclidean distance between point \mathbf{r} and atom center \mathbf{r}_i . The advantage of the RPC tessellations is that the resulting domains are guaranteed to be convex polyhedra¹³ whose inscribed spheres match the input radii $\{R_i\}$; see Fig. 1(b).

B. Physics-based definition of atomic size

To provide the weights (radii) for RPC, we must choose the “size” (radii) of each atom. A simple choice is the atomic radii from empirical or theoretical tables²⁷ which, however, is not site specific nor does it reflect bonding. For electronic-structure use, a judicious choice for each site (atomic or empty) are the minimum radii selected from the set of saddle-point radii (SPR) in the total electronic charge density, which reflect atomic “size.”¹⁵ Initiating a calculation, these SPR can be estimated by overlapping the isolated-atom charge densities in the desired structural positions, similar to Löwdin potentials.¹⁵ For a spherical-harmonic basis, we have shown that this SPR representation, even within atomic-sphere approximation, dramatically improves basis-set convergence and energetics in size-mismatched systems compared to full-potential methods.^{15,28} Site charges now also obey electronegativity rules, as found also with Bader topological (nonconvex) cells.^{15,28} Full details of applications are available in Ref. 15.

For RPC, given the site locations (structure) and SPR (weights), we use our modified version of Bernal’s FORTRAN software²⁹ to generate the VP information (vertices, faces, edges, etc.). Figure 2 shows the VP generated for a B2 cell with atoms of equal size and a tetragonally-distorted bct cell with atoms of unequal size.

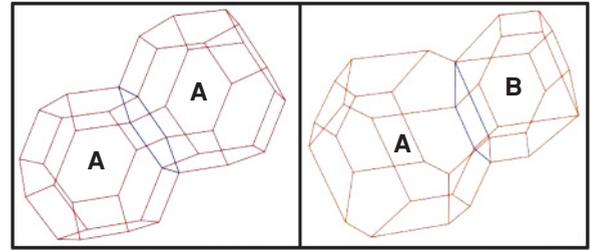


FIG. 2. (Color online) VP for two-atom unit cells with two inscribed radii. (left) B2 cell with equal radii each of type A, and (right) bct cell with unequal radii of type A and B.

C. Dual coordinate transformation and Gauss quadrature sums

Having divided the system into VP, there are two ways to proceed depending on the nature of the integrand $f(\mathbf{r})$. For simple integrands, separate each VP integration over a numerous simple polyhedra associated with each VP face and perform a Gauss quadrature sum, and the method works straightforwardly. If $f(\mathbf{r})$ has singularities near the origin, or if it is accessible only on a sparse grid, then two major VP subdomains need to be handled separately (i.e., inside and outside of the inscribed sphere). If $f(\mathbf{r})$ is spherical, the integral is one-dimensional and easy to perform accurately, whereas the second, interstitial domain is more challenging; see Fig. 3 for an fcc example.

The interstitial has too unusual a boundary for direct determination of suitable sampling points and their weights. To find the sampling points, we transform a bi-unit cube $-1 \leq x, y, z \leq 1$ into each piece of the interstitial formed by each VP face and the site center but cropped by the inscribed sphere. If any face has more than four vertices, points are added within the face (uniformly distributed) so that each face can be subdivided into polygons always having at most four vertices (a quadrilateral base); as a result, no interstitial subdomain has more than eight corners, like the cube. The same map used on the Gauss-Legendre points tells us the sampling positions in each interstitial subdomain. Note that one could use a triangular base, but we find that, while both subdivisions give the same results, the quadrilateral requires less operations; hence, it is more efficient.

For clarity, consider a one-atom fcc crystal, as in Fig. 3(a), where the VP consists of 12 quadrilateral faces which are

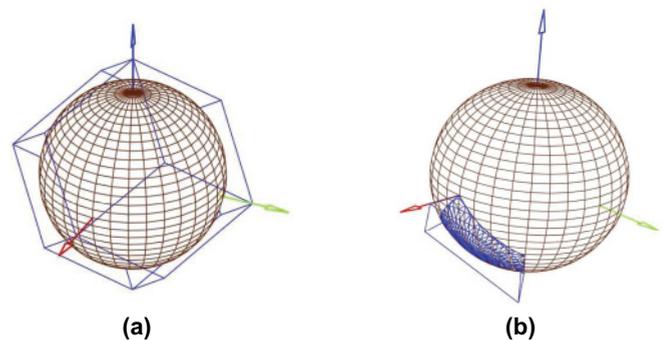


FIG. 3. (Color online) (a) VP of an fcc structure with twelve quadrilateral faces and an inscribed (touching) sphere. (b) A section of the VP shown as a single truncated pyramid.

divided into 12 cropped pyramids. Pick one, as in Fig. 3(b), and introduce spherical coordinates (r, θ, ϕ) so that the z axis is perpendicular to the VP face. Within each piece, the radius r runs from the inscribed radius R to the pyramid base (or VP face). To consider the case where the inscribed sphere integral is not done separately, take $R \rightarrow 0$ in what follows, and each pyramidal piece will no longer be cropped.

Before we map the cube to this element, we must find a transformation that flattens the curved interior surface. Choose any three of the four corner vertices formed by the intersection of the pyramid and the inscribed sphere. These three points are taken to define an interior plane. Now consider a cross section of the element at fixed angle ϕ or θ , which resembles Fig. 4(a). Note l_n is the distance from center of the inscribed sphere to the point of intersection of the radius vector with the interior plane. l_f is the distance to intersection with base plane (or face). Then the map

$$r = \frac{1}{l_f - l_n} [l_f(R - l_n) + r'(l_f - R)] \quad (1)$$

$$[x' \ y' \ z'] = \frac{1}{8} [1 \ x'' \ y'' \ z'' \ x''y'' \ y''z'' \ x''z'' \ x''y''z'']$$

will radially expand the interstitial piece (unprimed coordinates) so that the surface cut of the inscribed sphere will map to the interior plane (primed coordinates). Note that the map as given takes the plane to the sphere because, ultimately, we want a map from the cube to the interstitial piece. Despite the simplicity of the map [Eq. (1)], the Jacobian J_1 is nonpolynomial due to the angular dependence of $l_f(\theta, \phi)$ and $l_n(\theta, \phi)$. The standard determinant form of J_1 can be simplified by considering the volume change of an infinitesimal cell embedded in a spherical coordinate mesh. The cell will be stretched radially by a factor of $dr/dr' = (l_f - R)/(l_f - l_n)$. And, because the cell will be translated radially from r' to r , the base area will change from $r'^2 d\Omega$ to $r^2 d\Omega$. Thus, the total volume change (ratio) of the cell will be $\frac{(l_f - R)}{(l_f - l_n)} \frac{r^2}{r'^2}$.

Having flattened the interior curved surface, we then perform a second mapping from this hexahedra to a bi-unit $2 \times 2 \times 2$ cube, as depicted in Fig. 5. Let (x', y', z') and (x'', y'', z'') be the coordinates before and after the transformation, respectively. Mathematically, we can connect them using the expression

$$\begin{bmatrix} 1 & 1 & 1 & 1 & 1 & 1 & 1 & 1 \\ 1 & -1 & -1 & 1 & 1 & -1 & -1 & 1 \\ -1 & -1 & 1 & 1 & -1 & -1 & 1 & 1 \\ 1 & 1 & 1 & 1 & -1 & -1 & -1 & -1 \\ -1 & 1 & -1 & 1 & -1 & 1 & -1 & 1 \\ -1 & -1 & 1 & 1 & 1 & 1 & -1 & -1 \\ 1 & -1 & -1 & 1 & -1 & 1 & 1 & -1 \\ -1 & 1 & -1 & 1 & 1 & -1 & 1 & -1 \end{bmatrix} \begin{bmatrix} x'_1 & y'_1 & z'_1 \\ x'_2 & y'_2 & z'_2 \\ x'_3 & y'_3 & z'_3 \\ x'_4 & y'_4 & z'_4 \\ x'_5 & y'_5 & z'_5 \\ x'_6 & y'_6 & z'_6 \\ x'_7 & y'_7 & z'_7 \\ x'_8 & y'_8 & z'_8 \end{bmatrix}, \quad (2)$$

where the index in the subscript (1 to 8) indicates the vertex number in Fig. 5. In this map, we have reverted to describe the hexahedral element in cartesian coordinates (x', y', z') rather than the spherical (r', θ', ϕ') .

The Jacobian of the transformation J_2 that turns the hexahedra into a bi-unit cube is

$$J_2 = \begin{vmatrix} \frac{\partial x'}{\partial x''} & \frac{\partial x'}{\partial y''} & \frac{\partial x'}{\partial z''} \\ \frac{\partial y'}{\partial x''} & \frac{\partial y'}{\partial y''} & \frac{\partial y'}{\partial z''} \\ \frac{\partial z'}{\partial x''} & \frac{\partial z'}{\partial y''} & \frac{\partial z'}{\partial z''} \end{vmatrix}. \quad (3)$$

Thus, the volume integral over the interstitial region transforms to a volume integral over a cube. This can be expressed using Gaussian-Legendre integration as

$$\begin{aligned} \int_{\Omega^{IS}} f(\mathbf{r}) d^3 \mathbf{r} &= \int_{-1}^1 \int_{-1}^1 \int_{-1}^1 d^3 \mathbf{r}'' f(\mathbf{r}'') J_1 J_2 \\ &= \sum_{l=1}^{N_l} \sum_{m=1}^{N_m} \sum_{n=1}^{N_n} f(x''_l, y''_m, z''_n) J(x''_l, y''_m, z''_n) \\ &\quad \times w_l(x''_l) w_m(y''_m) w_n(z''_n), \end{aligned} \quad (4)$$

where $J = J_1 J_2$, and N_l , N_m , and N_n are the number of quadrature points along x'' , y'' , and z'' axes, respectively. The

Gauss points x_i and weights w_i are known analytically from the zeros of the Legendre polynomial, so Eq. (4) is straightforward to evaluate. Calculation time is primarily spent in numerically evaluating the analytically derived Jacobians J_1 and J_2 for the two successive transformations and the $f(x'', y'', z'')$; hence, quite fast. This isoparametric approach achieves machine-precision error for VP integrals involving volume, charge-densities and potentials. The function $f(x'', y'', z'')$ should be evaluated at the specified x_i points; if, however, f is only

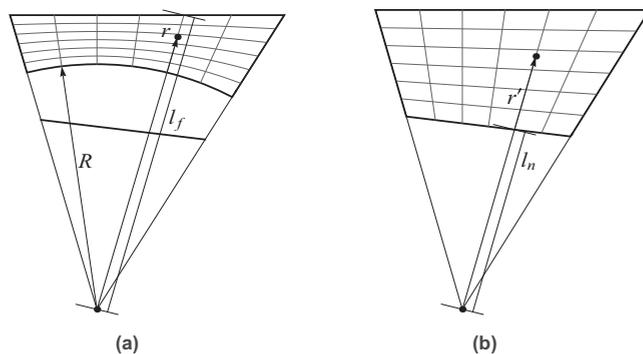


FIG. 4. Cross section of the cropped pyramid (a) before radial scaling and (b) after radial scaling.

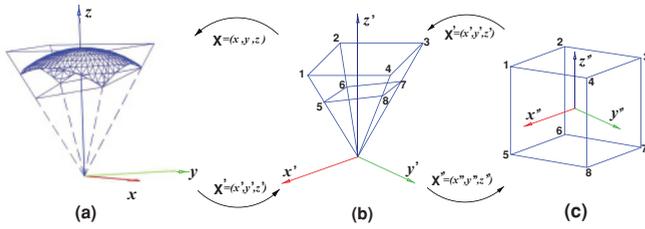


FIG. 5. (Color online) Two-step coordinate transformation: (1) Bottom curved surface (a) to the interior plane (b) via the Jacobian J_1 , and (2) hexahedra in (b) to the isoparametric $(2 \times 2 \times 2)$ bi-unit cube (c) via the Jacobian J_2 .

defined on a discrete grid, the function must be interpolated to each x_i , in which case interpolation error is the major error that should be ameliorated to achieve high-accuracy integration. Generally, if $f(x'', y'', z'')$ is a polynomial of order p_1 , p_2 , and p_3 along the three directions, respectively, then the number of sampling points N required to integrate the quantity exactly for a simple polyhedra domain is $(\frac{p_1}{2} + 1) \times (\frac{p_2}{2} + 1) \times (\frac{p_3}{2} + 1)$. For the case where we separate the integral over the inscribed sphere and integrate the interstitial over a domain that is curved, the transformation makes the integrand effectively nonpolynomial; therefore, more Gauss points will be required.

Our method is distinguished from that in Refs. 3 and 5 by the choice of transformations as well as partitioning space via the weighted VP. We transform the Gauss-Legendre sampling points inside a bi-unit cube into the truncated pyramid by (1) cubic polynomial mapping of the corner points of the cube to the corner points of the truncated pyramid (given by J_2), and then (2) by performing a linear mapping (in radius) of the interior plane (or side closest to origin) onto the relevant cut of the inscribed sphere (given by J_1). Our Jacobian $J \equiv J_1 J_2$ is always smooth and well behaved, even for a highly skewed pyramid. Baerends *et al.*^{3,5} have noted that their choice of coordinates can cause their intermediate functions to behave poorly (i.e., the J diverges) when the pyramid has wide opening angles, or a strongly skewed face. For our J to diverge, the interior plane would need to (nearly) touch the pyramid base plane; but, with the interior plane defined as the one passing through three of the intersection points of the inscribed sphere and the edges of the pyramid, this could only happen if the sphere touched one of the corners of the VP, which can never happen. In addition, the present procedure requires minimally fewer function evaluations.

D. Symmetry considerations

We could take advantage of the symmetry of the VP and crystal. We consider two kinds of symmetry. First, the point-group symmetry of the crystal structure identifies the set of inequivalent sites in the cell, which reduces computational time to that over inequivalent sites only. The second symmetry is associated with individual polyhedra. Because each polyhedra consists of various quadrilaterals associated with their faces, we can identify the symmetry-equivalent pyramids by applying a set of symmetry operators over each polyhedra around the symmetry-unique atoms. By integrating only over symmetry-inequivalent pyramids corresponding to each symmetry-inequivalent atom and weighting them with

their degeneracy, an appreciable savings in computer time would be obtained for systems with high symmetry. For example, there are 12 facets for VP in an elemental fcc structure so, at a minimum, we could perform an integration over one VP facet and multiply the result by 12; however, because each facet is four-fold symmetric, we could do 1/4 of the VP facet and multiply the result by 48, reducing VP integration by 1/48 of above integrations timings.

E. All-electron implementation

Our method is conceptually simple and easy to implement in any general electronic-structure code, with additional advantages for site-centered methods. For a spherical-harmonic basis, integrating separately over the inscribed sphere directly eliminates the Coulomb singularities due to the Jacobian within spherical coordinates. Also, using the optimal SPR basis¹⁵ we have better basis-set convergences and site charges. We have included this isoparametric method in a small set of RPC routines to determine the VP (vertices, faces, and edges). For complex cells, our algorithm has the flexibility to control the desired precision to balance the computational cost.

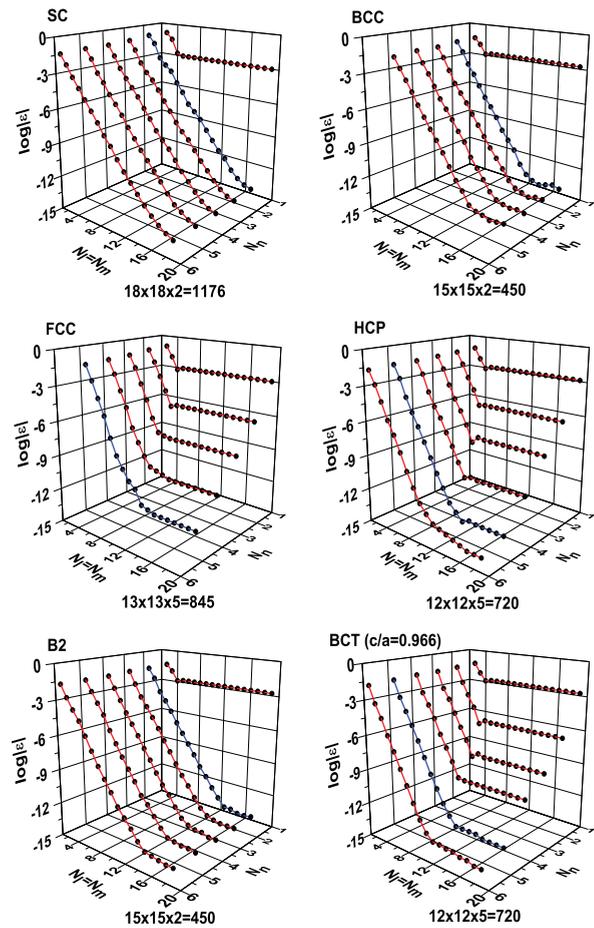


FIG. 6. (Color online) Logarithmic (base 10) error in interstitial volumes for six structures. N_i ($i = l, m, n$) is the number of Gauss points along \hat{x}'' , \hat{y}'' , and \hat{z}'' , respectively, with $N_n < N_l = N_m$ due to a smaller caliper along \hat{z}'' . Dark (blue) lines indicate minimum number of points along \hat{z}'' (total points listed below plots) to achieve 13 decimal accuracy.

This software is used to implement the (un)weighted VP-based isoparametric integration in our all-electron, KKR-CPA code.¹⁷ We discuss these results in Sec. V. Details of the calculations are as follows: The Green's function and related integrals use an external angular-momentum cutoff up to $L_{\max} = 3$ (i.e., s , p , d , and f symmetries), as needed. Energy integrations of the Green's functions are done by contour integration³⁰ via Gauss-Chebyshev methods with 18 energy points. We use the local spin-density approximation (LSDA) as parametrized by von Barth and Hedin.³¹ The Brillouin zone integrations use the Monkhorst and Pack³² special k -point method with 20^3 points in the full zone. For disordered alloys, we use the coherent-potential approximation³³ (CPA) based on the screened CPA³⁴ to incorporate more properly the metallic screening due to charge correlations in the local chemical environment.

IV. AN EXACTLY SOLVABLE MODEL

To illustrate the numerical convergence and accuracy, we use van Morgan's exactly solvable charge-density model.¹⁶ Many standard electronic-structure kernels can be exactly evaluated for the van Morgan density and potential, so the error in the numerical integrals can be precisely determined. We verify that accurate results are found with a modest number of Gauss points that depend on structure, and machine-precision can be achieved by increased number of points, slightly increasing computational time.

We showcase the convergence of volume and charge conservation, the $[\rho(\mathbf{r})V(\mathbf{r})]$ integral evaluated for kinetic and/or Coulomb energy, and more highly varying functions in l and \mathbf{r} . Apart from the cubic structures, we have also tested the convergence of the interstitial volume integral for more complex crystal structures. In the timings below, we have not utilized the associated symmetry of the crystal and the VP, so that the results reflect the most inequivalent case.

A. The van Morgan test problem

The van Morgan¹⁶ test charge density is defined as

$$\rho(\mathbf{r}) = B \sum_{n=1}^K e^{i \mathbf{T}_n \cdot \mathbf{r}}, \quad (5)$$

where \mathbf{T}_n are the nearest-neighbor reciprocal lattice vectors, and B is a scale factor. We will take $B = 1$ for simplicity. (From the Bauer expansion, a plane wave requires, in principle, an infinite number of spherical harmonics to be fully represented.) Because Ω^{VP} and Ω^{MT} are known exactly for any crystal structure, it is often convenient, especially for site-centered methods, to divide the VP into two volumetric regions: the volume of inscribed sphere Ω^{IS} and the volume within the interstitial region Ω^{IS} , so that $\Omega^{\text{VP}} = \Omega^{\text{IS}} \cup \Omega^{\text{MT}}$.

First, we can precisely assess the numerical error associated with volume conservation via

$$\int_{\text{IS}} d^3r = \Omega^{\text{IS}} = \Omega^{\text{VP}} - \Omega^{\text{MT}}, \quad (6)$$

where $\Omega^{\text{MT}} = 4\pi R^3/3$, and, for example, R is $1/2$, $\sqrt{2}/4$, and $\sqrt{3}/4$ for sc, fcc, and bcc (in units of lattice constant), respectively. The left-hand-side numerical integral is compared with the analytical result available for the right-hand side. For

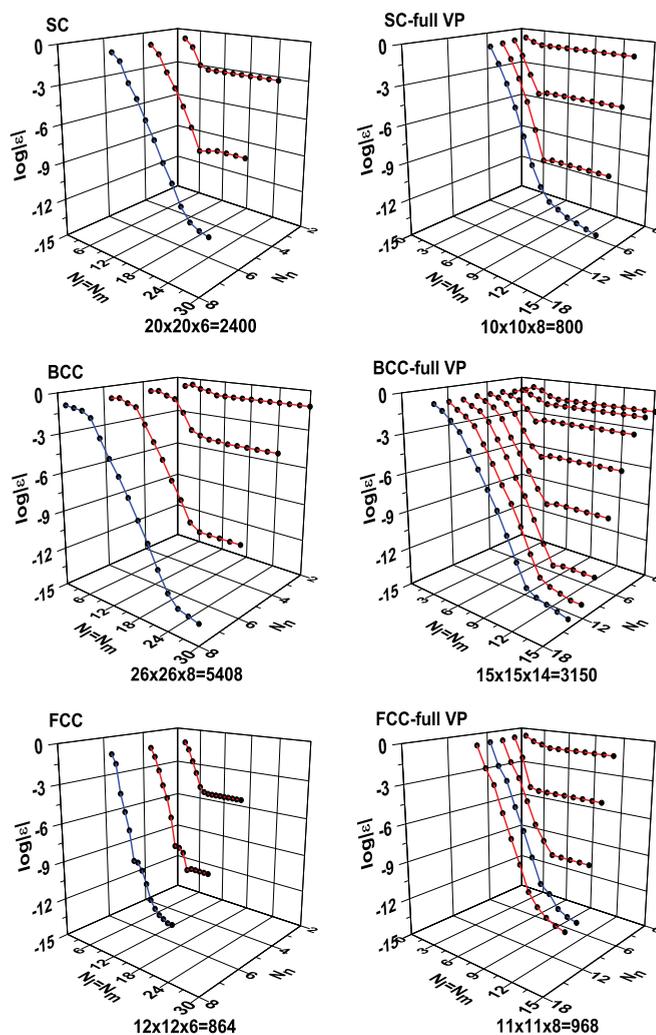


FIG. 7. (Color online) For the van Morgan problem for sc, bcc, and fcc, (left) the logarithmic (base 10) error in the interstitial charge [i.e., $\epsilon = (Q_{\text{calc}}^{\text{IS}} - Q_{\text{exact}}^{\text{IS}})/Q_{\text{exact}}^{\text{IS}}$] and (right) absolute error in VP total charge (i.e., $\epsilon = Q_{\text{calc}}^{\text{VP}} - Q_{\text{exact}}^{\text{VP}}$). Other details are as in Fig. 6.

example, the VP volumes are 1, 1/4, and 1/2 (in units of lattice constant cubed) for sc, fcc, and bcc, respectively.

Second, we can assess the integrations associated with charge conservation, including the determination of electronic chemical potential or Fermi energy. With $\rho(\mathbf{r})$ having no zero-mode component in its Fourier expansion, the integral of charge over a VP cell must be identically zero; hence, charge neutrality requires that

$$Q^{\text{total}} = \int_{\text{VP}} \rho(\mathbf{r}) d^3r = 0. \quad (7)$$

Subdivision of the VP yields

$$Q^{\text{IS}} = \int_{\Omega^{\text{IS}}} \rho(\mathbf{r}) d^3r = - \int_{\Omega^{\text{MT}}} \rho(\mathbf{r}) d^3r. \quad (8)$$

Next, we can assess numerical errors for the $\rho(\mathbf{r})V(\mathbf{r})$ integral, which can be expressed as

$$\begin{aligned} [\rho V]^{\text{IS}} &= \int_{\Omega^{\text{IS}}} \rho(\mathbf{r}) V(\mathbf{r}) d^3r \\ &= \frac{4\pi K \Omega^{\text{IS}}}{|T_n|^2} - \int_{\Omega^{\text{MT}}} \rho(\mathbf{r}) V(\mathbf{r}) d^3r. \end{aligned} \quad (9)$$

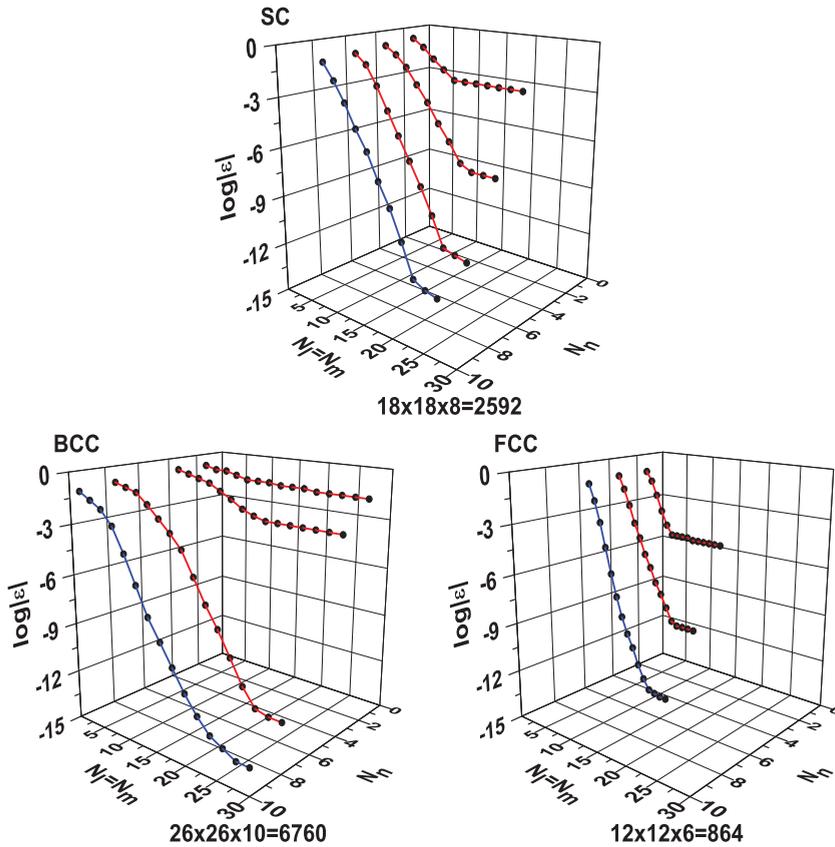


FIG. 8. (Color online) For the van Morgan problem for sc, bcc, and fcc, the logarithmic (base 10) error in the interstitial $[\rho V]$ integral. Other details are as in Fig. 6.

The exact analytic solutions of the above kernels for three cubic structures (sc, fcc and bcc) are given in Appendix. However, noncubic structures are numerically no more difficult or error prone than these cubic cases (but they cannot be performed analytically). Besides band energy (an eigenvalue summation requiring a Fermi energy) and exchange correlation, the above three integrals reflect the main integrations contributing to DFT total energies, for example.

B. Complex varying integrands

The general method we have presented here can integrate over any arbitrary polyhedra for any complicated function, such as those with high angular momentum or those that are strongly varying with(out) exponential decay. Here we showcase a set of strongly varying integrands that are critical for evaluating the near-field contributions to the Poisson equation (in the so-called moon region) for site-centered, electronic-structure methods;³⁵ namely,

$$a_{lm} = \sum_{\mathbf{R} \neq 0} \int_{\text{VP}} d\mathbf{r}' \rho_{\mathbf{R}}(\mathbf{r}') \frac{Y_{lm}(\widehat{\mathbf{r}' + \mathbf{R}})}{|\mathbf{r}' + \mathbf{R}|^{l+1}}, \quad (10)$$

where Y_{lm} are the spherical harmonics and there is a Madelung summation over \mathbf{R} . It is a rapidly decaying function with increasing l and, to achieve high precision of this piece of the Coulomb potential, rather high values of l are required. Convergence of the above integral for various $\{lm\}$ values is shown in the next section. We have also tried other more strongly varying functions and again achieved accurate results with a modest number of Gauss points. As will be discussed elsewhere, most codes that implement the correct Poisson

solution for space-filling VP cannot do the integrals for skewed VP, or they are not accurate enough due to use of, for example, shape functions (an example appears below).

V. RESULTS AND DISCUSSION

A. Accuracy

To illustrate the convergence of isoparametric integration, Fig. 6 shows the logarithmic error in interstitial volume for six structures (i.e., 1-atom cubics, 2-atom hcp, and 2-atom B2 and bct). Each point on the graph represents the result for a combination of quadrature points (N_l, N_m, N_n) . From Fig. 5(a), it is clear that the cropped pyramid has a thinner dimension along the z axis compared to the other two axes. Therefore, we use less quadrature points along \hat{z}'' than along \hat{x}'' and \hat{y}'' [i.e., $N_n < (N_l, N_m)$]; in particular, we used $N_l = N_m$. An accuracy of around 10^{-3} is already reached with only $N_l = N_m = 4$ points along \hat{x}'' and \hat{y}'' . The darker line in each panel shows the minimum number of quadrature points along \hat{z}'' to achieve convergence to 13 decimal places. For example, the minimum number of Gauss points along \hat{z}'' for a bcc structure to attain an error less than 10^{-13} is two. The minimum number of points (N_l, N_m, N_n) required is listed below each subpanel.

The convergence of the charge density integral (Q) is given in Fig. 7. The left panel shows the logarithmic error in the interstitial charge Q^{IS} for the cubic structures. The right panel shows the absolute error $\epsilon^{\text{VP}} = Q_{\text{calc}}^{\text{VP}} - Q_{\text{exact}}^{\text{VP}}$ in the total charge integral. The charge convergence requires more points to yield a similar level of accuracy. For example, to achieve an accuracy of up to the third decimal place, the bcc structure requires 8 points along \hat{x}'' and \hat{y}'' compared to the 4

points needed for the sc and fcc structures. Higher accuracy requires more points for the bcc case due to its wider and more asymmetric interstitial region.

Figure 8 shows the convergence of the interstitial $[\rho V]$ integral for the cubic structures. It is interesting to notice that the $[\rho V]$ integral converges almost at the same rate as the ρ integral (left panel of Fig. 7) and does not take any longer for all three structures. The bcc structure in this case also requires comparatively more points to converge for the same reason as in the previous paragraph. We have analyzed other relevant integrands (often used in electronic-structure calculations) as well and found either a similar or a slightly slower rate of convergence.

In Table I, we list the minimum number of points required to get the interstitial volume, charge, and $[\rho V]$ -integral convergence to more than the 13th decimal for each structure. The number of points required are given as $\{N_l = N_m, N_n\}$. Table II shows similar results for the full VP integral of strongly varying functions in Eq. (10) for various $\{l, m\}$ values, exhibiting oscillatory angular dependence with l -dependent spatial decay. In spite of its strongly varying nature, the number of Gauss points $\{N\}$ required to achieve an accuracy of up to 10 decimal places is not large and is comparable to that of the ρ and ρV integrals shown in Table I.

The accuracy of all our integrals is limited by the accuracy of the VP boundary (vertices, faces, and edges) information generated from Bernal's software.²⁹ We have modified Bernal's original (binary-math, single-precision) code to improve its efficiency and extend its accuracy, and we were able to achieve just below 10^{-13} . We have verified that our main limitation in accuracy is due to lack of a double-precision real code. By rewriting the software from scratch, which is a considerable effort beyond the scope of the present work, we could certainly achieve machine precision. Therefore, all integration results will be limited to just below 10^{-13} ; with improved accuracy of VP information, machine-precision is achievable with similar Gauss points described.

B. Efficiency

To contrast the VP construction timings, we compare to the time required to expand the shape function (or three-dimensional step function) into spherical harmonics.³⁶ The shape-function approach is often used in the community when needing site-dependent quantities. The EMTO, KKR, LSMS, APW, etc. methods, for example, typically report site-quantities, and KKR Green's function methods require site-dependent VP scattering matrices.

The shape-truncated function for a VP is defined as

$$\sigma(\mathbf{r}) = \begin{cases} 1, & \mathbf{r} \in \Omega \\ 0, & \mathbf{r} \notin \Omega, \end{cases} \quad (11)$$

where Ω is the VP region. The expansion of $\sigma(\mathbf{r})$ in spherical harmonics yields the angular momentum decomposition

$$\sigma_L(|\mathbf{r}|) = \int_{\hat{\mathbf{r}}} d\hat{\mathbf{r}} Y_L^*(\hat{\mathbf{r}}) \sigma(\mathbf{r}) \equiv \sigma_L(r), \quad (12)$$

where the integration is over the angles $\hat{\mathbf{r}} \equiv (\theta, \phi)$ and $L \equiv (l, m)$. The shape function is used to simplify the numerical

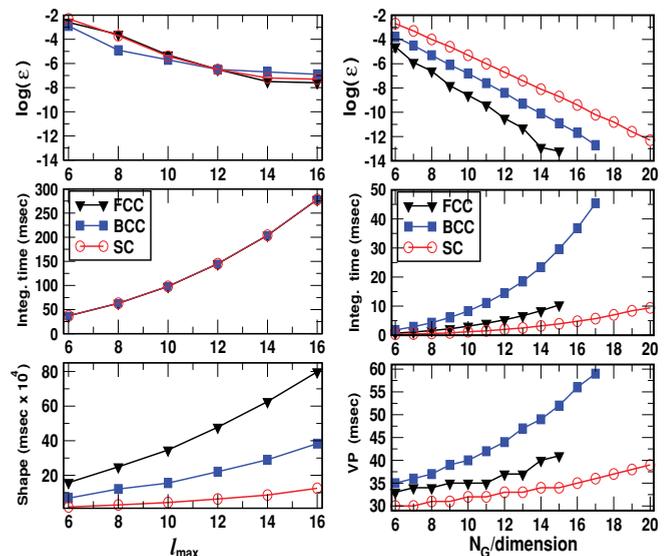


FIG. 9. (Color online) Timings to achieve a specific level of interstitial-charge accuracy for cubic structures using shape function (left) versus isoparametric (right) integration. Shown in panels is the logarithmic error in the interstitial charge (top), and the times to construct VP boundary information (bottom) and to integrate (middle). Isoparametric integration is $>10^5$ times and achieves machine precision.

integration of any function $f(\mathbf{r})$ over the polyhedron volume Ω as

$$\begin{aligned} F &= \int_{\Omega} f(\mathbf{r}) \sigma(\mathbf{r}) d^3r \\ &= \sum_{L=0}^{L_{\max}} \int dr r^2 \sigma_L(r) \int_{\Omega} d\Omega Y_L(\hat{\mathbf{r}}) f(\mathbf{r}), \end{aligned} \quad (13)$$

especially if it is well represented by spherical harmonics.

The expansion coefficients $\sigma_L(r)$ must be truncated at a very high $L_{\text{trunc}} \gg L_{\max}$ to achieve an accurate representation of the VP shape and to obtain a reliable integral value. For example, for an fcc structure, $\rho(\mathbf{r})$ is well represented using $L \leq 8$ (i.e., $L_{\max} = 8$), but the shape-function should have $L_{\text{trunc}} \gg 4L_{\max}$ to have converged $\sigma_{L \leq 8}(r)$ that will yield an accurate integral. As we shall see, this L_{trunc} will limit the accuracy of the integrals in the codes that use this approach, making the shape-function approach unacceptable for general (non-high-

TABLE I. Convergence for the interstitial volume, charge, and $[\rho V]$ integrals for various crystal structures. $\{N_l = N_m, N_n\}$ are the optimal number of points for each structure to reach an accuracy of at least 13 decimal places. VC, QC and $[\rho V]$ stands for the volume, charge and $[\rho V]$ -integral convergence.

Structure	$\{N_l, N_n\}_{VC}$	$\{N_l, N_n\}_{QC}$	$\{N_l, N_n\}_{\rho V}$
sc	{18,2}	{20,6}	{18,8}
bcc	{15,2}	{26,8}	{26,10}
fcc	{13,5}	{12,6}	{12,6}
hcp	{12,5}		
B2	{15,2}		
bct	{12,5}		

symmetry) structures, where L_{trunc} should be significantly larger than in the cubic cases to achieve the same level of accuracy as for fcc.

Figure 9 shows accuracy and computer time for isoparametric (right panel) and shape-function (left panel) methods for sc, fcc, and bcc, for a direct comparison. The rate of convergence is given with respect to the number of Gauss points along each dimension for the present method and with respect to l_{max} for a fixed radial grid using shape functions. The present method attains error in the van Morgan interstitial charge below 10^{-13} with less computational time. The shape-function technique cannot achieve an accuracy better than 10^{-7} with $l_{\text{max}} = 16$; an extremely expensive calculation due to the high- L expansion. Hence, our method provides some significant advantages over existing approaches.

The bottom panel shows the time required to generate the boundary information necessary to achieve a certain level of accuracy. For both methods, most of the time is spent in determining the VP boundaries. The present method generates this information in terms of neighbors, vertices, faces, and edges for each VP. The shape-function method gets the VP shape in terms of an L expansion on a specific radial grid. Clearly, the shape-function method requires $>10^4$ times more time than the present method. The middle panel shows the time (in ms) required to sum the final expression for the integration for both VP or shape function. The present method is faster by >7 times. Overall, using no symmetry (degeneracy) information to reduce the computational time, we achieve $\simeq 10^5$ times faster integration with 10^6 times less error.

C. Application to FePd

We present the formation enthalpy, ΔE_f , for fcc-based ferromagnetic (FM) (dis)ordered Fe-50%Pd; that is, A1 solid-solution and ordered $L1_0$, from an unweighted (equal) VP and SPR-weighted VP, giving an optimal partitioning of space for integration and a concomitantly improved basis set, as

TABLE II. Convergence of fcc a_{lm} in Eq. (10) versus $\{l, m\}$ (R is summed to 8th neighbor shell). $\{N\}$ is the number of points per direction for 10 decimal place accuracy.

l	m	$\{N\}_{a_L}$	$[a_{lm}]_{\text{numerical}}$	$[a_{lm}]_{\text{exact}}$
0	0	12	0.009 951 109 455	0.009 951 109 341
2	0	12	0.000 000 000 000	0.000 000 000 000
4	0	14	-9.449 717 387 589	-9.449 717 387 292
4	4	14	-5.647 286 285 886	-5.647 286 285 399
6	0	16	-10.626 482 313 14	-10.626 482 313 81
6	4	16	19.880 328 021 19	19.880 328 021 74
8	0	18	65.840 245 144 25	65.840 245 144 10
8	4	18	24.759 271 364 01	24.759 271 364 34
8	8	19	37.723 807 706 70	37.723 807 706 99
10	0	21	135.852 078 523 4	135.852 078 523 9
10	4	21	-136.893 105 843 2	-136.893 105 843 8
10	8	24	-162.935 386 290 1	-162.935 386 292 8
12	0	24	-205.205 030 788 5	-205.205 030 787 8
12	4	26	-493.649 392 083 8	-493.649 392 082 9
12	8	26	544.811 362 712 9	544.811 362 713 5
12	12	26	-261.804 641 588 3	-261.804 641 588 3

described in Sec. III B and shown in Fig. 2. Formation energies are highly relevant for phase stability; see Ref. 28 for an example application to phase stability of magnetic-storage materials.

Table III shows the ΔE_f for FM $L1_0$ and A1 phases versus the external L_{max} for the local spherical basis. Using the weighted-VP integration, results become significantly less sensitive to L_{max} , which is especially clear for the energy difference ΔE^{o-d} between ordered and disordered phases, which remains almost constant, in contrast to the unweighted case. Our weighted integration yields formation energetics in very good agreement with that observed for $L1_0$.³⁷ The weighted-VP integration thus provides accurate results, a minimal basis set in terms of angular momentum cutoff, and a significant reduction in matrix-inversion time because of the now-permitted use of the lower rank of the KKR matrices $N(L_{\text{max}} + 1)^2$, where N is the number of atoms in the unit cell.

We could not find enthalpy data for the FM-A1 phase. Therefore, we provide $\Delta E_{\text{PM}}^{o-d}$ in the paramagnetic phase (via disordered local moment state), which is related to the order-disorder temperature.²⁸ Indeed, for the weighted-VP case, $\Delta E_{\text{PM}}^{o-d}$ is 82 meV (or 952 K), which is close to the order-disorder temperature of 1050 K³⁸ and shows that the disordered phase results are now accurate, too.

Large-scale boundaries are critical in materials science; for example, for mechanical properties as pinning centers for mageto-anisotropy. As a test of the present weighted-VP integration, we calculated the [001] antiphase boundary planar defect energy of $L1_0$ -FePd, a long-period structure with 32 atoms per cell with varying interstitial regions. We find 910 mJ/m² versus 890 mJ/m² from VASP plane-wave calculations, only ours is about 20 times faster and provides local properties directly.

In addition, the magnitude of site excess (or deficient) charge (i.e., the “charge transfer”) in a solid crucially depends on the way in which the space is divided into geometric cells. For space-filling VP, SPR-weighted cells will provide a more physics-based partitioning of space and more realistic assessment of charge transfer. Approximate methods like the popular atomic-sphere approximation (ASA) has an overlap error; the situation becomes worse for non-close-packed materials.

TABLE III. Formation enthalpy ΔE_f (in meV/atom) for (dis)ordered ferromagnetic FePd versus L_{max} from equal (unweighted) and SPR-weighted VP, along with ordering energies ΔE^{o-d} . SPR results are much less sensitive to basis-set L_{max} cutoff. KKR results are compared to other results.

L_{max}	Unweighted VP			Weighted VP		
	ΔE_f^{ord}	ΔE_f^{dis}	ΔE^{o-d}	ΔE_f^{ord}	ΔE_f^{dis}	ΔE^{o-d}
2	-10.7	+18.6	-29.3	-83.3	-59.4	-23.9
3	-45.7	+8.3	-54.0	-88.8	-63.9	-24.9
4	-30.7	-9.9	-20.8	-86.5	-61.7	-24.8
	Expt. (Ref. 37)			-98 ± 11		
	VASP-LDA (GGA)			+40(-130)		

TABLE IV. Excess charges within VP or ASA spheres for (dis)ordered FePd with (un)weighted VP via KKR.

Method	Unweighted		Weighted	
	ΔQ^{ord}	ΔQ^{dis}	ΔQ^{ord}	ΔQ^{dis}
VP Fe	-0.111	+0.059	-0.032	-0.026
Pd	+0.111	-0.059	+0.032	+0.026
ASA Fe	-0.139	+0.089	-0.082	-0.051
Pd	+0.139	-0.089	+0.082	+0.051

In Table IV, we show the calculated excess charges within (un)weighted VP sites in A1 and L1₀ FePd, with comparison to (un)weighted ASA spheres used in many popular codes. Reference¹⁵ provides details of the ASA approach. Generally, there is a charge transfer from Fe to Pd, as expected from the electronegativities.

However, for unweighted cases in the A1 phase, there is an excess charge on the small (Fe) atom, which is distinctly unphysical and is due to the tails of the charge density of large (Pd) atoms being improperly cut off at the smaller radii. When a weighted VP or ASA is used, this situation is corrected (the sign changes) because the charge density is now better represented in the disordered phase.¹⁵ For the unweighted L1₀ case, there is a large depletion of charge on the small (Fe) atom due to a Madelung effect; however, for the weighted case, the inscribed sphere reflects more appropriately the extent of the charge density and, hence, it is a more reliable estimate. Importantly, there is a large reduction in excess (depleted) charges for the weighted-VP integration compared to the weighted-ASA case (now with the correct sign), which shows the error arising from overlap of spheres.

VI. SUMMARY

We have presented a fast, accurate, and easy-to-implement method for numerical integration over general VP for polyatomic systems. The algorithm combines a weighted Voronoi partitioning of space with isoparametric integration using the Gauss-Legendre quadrature formulas of product type, and does not suffer from any ill behavior with VP shape. In contrast to other methods, accuracy and convergence was tested rigorously via an analytic charge-density model, with machine-precision accuracy for a reasonable number of Gauss points. We showed also that our algorithm is 10⁵ times faster and 10⁷ times more accurate than that based on shape functions used in several electronic-structure codes. Our method could be used for other types of condensed matter problems requiring integration over arbitrary convex VP. Here, we implemented the general method in a site-centered, electronic-structure code and calculated formation enthalpies for FePd, yielding good agreement with experiment. The radii to set the Voronoi-Delauney tessellation weights is obtained from a physics-based definition (i.e., the saddle points in the total electron density).

ACKNOWLEDGMENTS

Our work was supported by the U.S. Department of Energy BES/Materials Science and Engineering Division from contracts with Illinois (DEFG02-03ER46026), Ames Laboratory (DE-AC02-07CH11358), operated by Iowa State University, and Lawrence Livermore National Laboratory (subcontract B573247). DDJ and SNK acknowledge support from the NSF (DMR-07-05089) and, recently, the Center for Defect Physics, Energy Frontier Research Center. We thank Wan-Tsang Wang (NSYSU, Taiwan, and advisor Y.C. Chang) for help in algorithm testing while in Illinois.

APPENDIX: EXACT SOLUTION FOR CUBIC STRUCTURES

For volume conservation the VP and inscribed volume are known analytically; the interstitial integral is

$$\Omega_{\text{exact}}^{\text{IS}} = \Omega^{\text{VP}} - \frac{4\pi}{3} [R_{\text{MT}}^{(s)}]^3, \quad (\text{A1})$$

where (s) indicates the lattice type (sc, fcc, or bcc). $R_{\text{MT}}^{(s)}$ is the inscribed sphere or (muffin-tin) radius of the lattice s . For charge conservation, it is straightforward to show that

$$Q_{\text{exact}}^{\text{IS}} = \frac{A^{(s)}\pi}{[T^{(s)}]^3} [\sin \alpha^{(s)} - \alpha^{(s)} \cos \alpha^{(s)}], \quad (\text{A2})$$

where $\alpha^{(s)} = T^{(s)} R_{\text{MT}}^{(s)}$, $T^{(s)} \equiv |\mathbf{T}_n|$. $A^{(s)}$ is a normalization constant whose value is 24, 32, and 48 for the sc, fcc, and bcc lattice, respectively. Finally, the exact expression for the ρV integral (9) for the van Morgan charge and potential for any general lattice can be simplified for the cubic lattices (sc, fcc, and bcc) as

$$[\rho V]_{\text{exact}}^{\text{IS}} = \frac{4\pi K^{(s)}[\Omega^{\text{IS}}]^{(s)}}{[T^{(s)}]^2} - \frac{4\pi}{[T^{(s)}]^3} \sum_{i=1}^{p^{(s)}} f_i^{(s)} \times \left[\frac{\sin \gamma_i^{(s)} - \gamma_i^{(s)} \cos \gamma_i^{(s)}}{[\beta_i^{(s)}]^3} \right], \quad (\text{A3})$$

where $K^{(s)} = K = 6, 8, 12$ and $p^{(s)} = 2, 3, 4$ for the three lattices, respectively, and $\gamma_i^{(s)} = \beta_i^{(s)} \alpha^{(s)}$.

For the above integral expressions, the coefficients f_i and β_i for the sc, fcc, and bcc lattices are

$$\begin{aligned} \text{for sc, } & f_1 = \frac{f_2}{K-2} = K, \quad \beta_1 = \sqrt{2}\beta_2 = 2; \\ \text{for fcc, } & f_1 = \frac{2}{K-2}f_2 = \frac{2}{K-2}f_3 = K, \\ & \beta_1 = \sqrt{2}\beta_2 = \sqrt{\frac{3}{2}}\beta_3 = 2; \\ \text{for bcc, } & f_1 = \frac{2}{K-4}f_2 = \frac{2}{K-4}f_3 = \frac{1}{2}f_4 = K, \\ & \beta_1 = \frac{2}{\sqrt{3}}\beta_2 = 2\beta_3 = \sqrt{2}\beta_4 = 2. \end{aligned}$$

*ddj@ameslab.gov

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