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## Abstract

X-ray single crystal and powder diffraction studies on the  $\text{Gd}_5\text{Ga}_x\text{Ge}_{4-x}$  system with  $0 \leq x \leq 2.2$  reveal dependence of interslab T–T dimer distances and crystal structures themselves on valence electron concentration (T is a mixture of Ga and Ge atoms). While the  $\text{Gd}_5\text{Ga}_x\text{Ge}_{4-x}$  phases with  $0 \leq x \leq 0.6$  and valence electron concentration of 30.4–31  $e^-/\text{formula}$  crystallize with the  $\text{Sm}_5\text{Ge}_4$ -type structure, in which all interslab T–T dimers are broken (distances exceeding 3.4 Å), the phases with  $1 \leq x \leq 2.2$  and valence electron concentration of 28.8–30  $e^-/\text{formula}$  adopt the  $\text{Pu}_5\text{Rh}_4$ - or  $\text{Gd}_5\text{Si}_4$ -type structures with T–T dimers between the slabs. An orthorhombic  $\text{Pu}_5\text{Rh}_4$ -type structure, which is intermediate between the  $\text{Gd}_5\text{Si}_4$ - and  $\text{Sm}_5\text{Ge}_4$ -type structures, has been identified for the  $\text{Gd}_5\text{GaGe}_3$  composition. Tight-binding linear-muffin-tin-orbital calculations show that substitution of three-valent Ga by four-valent Ge leads to larger population of the antibonding states within the dimers and, thus, to dimer stretching and eventually to dimer cleavage.

## Disciplines

Materials Chemistry | Other Chemistry | Physical Chemistry

## Comments

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## Phase Transformation Driven by Valence Electron Concentration: Tuning Inter slab Bond Distances in $\text{Gd}_5\text{Ga}_x\text{Ge}_{4-x}$

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**Abstract:** X-ray single crystal and powder diffraction studies on the  $\text{Gd}_5\text{Ga}_x\text{Ge}_{4-x}$  system with  $0 \leq x \leq 2.2$  reveal dependence of inter slab T–T dimer distances and crystal structures themselves on valence electron concentration (T is a mixture of Ga and Ge atoms). While the  $\text{Gd}_5\text{Ga}_x\text{Ge}_{4-x}$  phases with  $0 \leq x \leq 0.6$  and valence electron concentration of 30.4–31 e<sup>-</sup>/formula crystallize with the  $\text{Sm}_5\text{Ge}_4$ -type structure, in which all inter slab T–T dimers are broken (distances exceeding 3.4 Å), the phases with  $1 \leq x \leq 2.2$  and valence electron concentration of 28.8–30 e<sup>-</sup>/formula adopt the  $\text{Pu}_5\text{Rh}_4$ - or  $\text{Gd}_5\text{Si}_4$ -type structures with T–T dimers between the slabs. An orthorhombic  $\text{Pu}_5\text{Rh}_4$ -type structure, which is intermediate between the  $\text{Gd}_5\text{Si}_4$ - and  $\text{Sm}_5\text{Ge}_4$ -type structures, has been identified for the  $\text{Gd}_5\text{GaGe}_3$  composition. Tight-binding linear-muffin-tin-orbital calculations show that substitution of three-valent Ga by four-valent Ge leads to larger population of the antibonding states within the dimers and, thus, to dimer stretching and eventually to dimer cleavage.

### Introduction

Recent discovery of the giant magnetocaloric effect in  $\text{Gd}_5\text{Si}_2\text{Ge}_2$ <sup>1–3</sup> triggered extensive research in the  $\text{R}_5\text{X}_4$  systems (R is a rare-earth element and X is a main group element).<sup>4–16</sup> Interest in these materials is fueled by economic benefits, i.e., potential application of  $\text{Gd}_5\text{Si}_2\text{Ge}_2$ <sup>17–20</sup> for room-temperature

magnetic refrigeration with larger efficiency than current vapor-cycle units, as well as by scientific curiosity, directed toward understanding this unusual phenomenon.<sup>21–24</sup> The magnetic ordering in  $\text{Gd}_5\text{Si}_2\text{Ge}_2$  is coupled with a reversible, first-order structural transformation: the low-temperature ferromagnetic form adopts an orthorhombic  $\text{Gd}_5\text{Si}_4$ -type structure with T–T dimers between  $\infty^2[\text{Gd}_5\text{T}_4]$  slabs (T is a statistical mixture of Ge and Si atoms on the corresponding sites), and the room-temperature paramagnetic form has a monoclinic  $\text{Gd}_5\text{Si}_2\text{Ge}_2$ -type structure, in which half of the T–T inter slab dimers are broken.<sup>21</sup> This magnetic/martensitic transition can be controlled by changing composition, temperature, pressure, and magnetic field.<sup>21,23,25</sup>

One of the interesting features of the transformation is that the low-temperature phase has a higher symmetry ( $Pnma$  space group) than the room-temperature phase ( $P112_1/a$  space group). Calculations by Choe et al.<sup>21</sup> and later by Pecharsky et al.<sup>26</sup>

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**Table 1.** Crystallographic Data for  $Gd_5Ga_xGe_{4-x}$  Powders and Single Crystals

x	sample	str. type of phases	a, Å	b, Å	c, Å	c/a	V, Å <sup>3</sup>
0	powder	Sm <sub>5</sub> Ge <sub>4</sub>	7.6939(3)	14.8232(6)	7.7825(3)	1.01152(5)	887.6(1)
	crystal	Sm <sub>5</sub> Ge <sub>4</sub>	7.683(2)	14.811(4)	7.774(2)	1.0118(3)	884.6(4)
0.5	powder	Sm <sub>5</sub> Ge <sub>4</sub>	7.679(6)	14.87(1)	7.808(4)	1.0168(9)	892(1)
	crystal	Sm <sub>5</sub> Ge <sub>4</sub>	7.660(1)	14.860(3)	7.811(2)	1.0197(3)	889.2(3)
0.6	powder	Sm <sub>5</sub> Ge <sub>4</sub>	7.6598(6)	14.879(1)	7.8141(5)	1.0201(1)	890.6(1)
0.7	powder <sup>a</sup>	Sm <sub>5</sub> Ge <sub>4</sub> + Pu <sub>5</sub> Rh <sub>4</sub>	7.563(7)	14.88(1)	7.88(1)	1.042(2)	887(1)
	crystal	Sm <sub>5</sub> Ge <sub>4</sub> <sup>b</sup> + Pu <sub>5</sub> Rh <sub>4</sub>	7.6195(8)	14.838(2)	7.8162(8)	1.0258(3)	883.7(2)
0.8	powder <sup>a</sup>	Sm <sub>5</sub> Ge <sub>4</sub> + Pu <sub>5</sub> Rh <sub>4</sub>	7.5518(5)	14.896(4)	7.8894(6)	1.0447(1)	887.5(3)
	crystal	Sm <sub>5</sub> Ge <sub>4</sub> <sup>b</sup> + Pu <sub>5</sub> Rh <sub>4</sub>	7.613(1)	14.880(3)	7.846(2)	1.0306(3)	888.7(3)
0.9	powder <sup>a</sup>	Sm <sub>5</sub> Ge <sub>4</sub> + Pu <sub>5</sub> Rh <sub>4</sub>	7.5438(3)	14.892(8)	7.8827(4)	1.04492(7)	885.6(5)
1	powder	Pu <sub>5</sub> Rh <sub>4</sub>	7.5473(2)	14.9217(9)	7.8951(5)	1.04608(9)	889.1(1)
	crystal	Pu <sub>5</sub> Rh <sub>4</sub>	7.572(2)	14.933(3)	7.884(2)	1.0412(3)	891.5(3)
1.2	powder	Gd <sub>5</sub> Si <sub>4</sub>	7.5348(3)	14.9485(4)	7.9046(4)	1.04908(7)	890.3(1)
1.5	powder	Gd <sub>5</sub> Si <sub>4</sub>	7.5234(3)	14.9892(6)	7.9183(3)	1.05249(6)	892.9(1)
2	powder	Gd <sub>5</sub> Si <sub>4</sub>	7.5226(4)	15.0068(8)	7.9275(4)	1.05382(8)	894.9(1)
	crystal	Gd <sub>5</sub> Si <sub>4</sub>	7.5162(7)	14.971(1)	7.9149(7)	1.0531(1)	890.6(1)
2.2	powder	Gd <sub>5</sub> Si <sub>4</sub>	7.5176(4)	15.0111(9)	7.9260(5)	1.05433(9)	894.4(1)

<sup>a</sup> Lattice parameters of the Pu<sub>5</sub>Rh<sub>4</sub>-type phases are given. <sup>b</sup> Dominant phase.

have shown that this unusual phenomenon arises from the large magnetic exchange coupling, which is optimized in the orthorhombic phase due to a higher valence electron concentration available for metallic bonding. The studies linked cleavage of the T–T interslab bonds and, thereby, the structure of Gd<sub>5</sub>Si<sub>2</sub>Ge<sub>2</sub> to the number of valence electrons in the conduction band. However, this dependence is not clear-cut due to the fact that the distortion is temperature dependent and is accompanied by magnetic ordering. Moreover, while increasing the Ge amount in Gd<sub>5</sub>Si<sub>2</sub>Ge<sub>2</sub> without changing the electron concentration (Si and Ge are isoelectronic) results in complete breaking of the remaining T–T interslab dimers at room temperature and suppresses the coupled magnetic and structural transitions to much lower temperatures, raising the Si concentration eliminates the structural transition entirely through stabilizing the Gd<sub>5</sub>Si<sub>4</sub>-type structure throughout the whole temperature range.<sup>23</sup> In this light, a system, which can unambiguously correlate a structure to an electron concentration, was highly desirable to get a better understanding of the symmetry-breaking process in Gd<sub>5</sub>Si<sub>2</sub>Ge<sub>2</sub> and other related phases. Because Gd<sub>5</sub>Ge<sub>4</sub> adopts two structures, a low-temperature, field-induced ferromagnetic one (Gd<sub>5</sub>Si<sub>4</sub>-type) with all interslab T–T dimers intact and a high-temperature paramagnetic one (Sm<sub>5</sub>Ge<sub>4</sub>-type) with all T–T dimers broken,<sup>27</sup> substituting three-valent, size-equivalent Ga (metallic radius,  $r_m$ , 1.246 Å) for four-valent Ge ( $r_m$  = 1.242 Å) in Gd<sub>5</sub>Ge<sub>4</sub> could tune the interslab bonds and, thereby, induce a phase transformation through a change in the valence electron concentration. In this paper, we report on structural variations in the Gd<sub>5</sub>Ga<sub>x</sub>Ge<sub>4-x</sub> system for  $0 \leq x \leq 2.2$ . Surprisingly, by tuning the valence electron concentration, we find not only a structural transformation from the Sm<sub>5</sub>Ge<sub>4</sub> type to the Gd<sub>5</sub>Si<sub>4</sub> one, but also a new intermediate structure, namely the orthorhombic Pu<sub>5</sub>Rh<sub>4</sub> one,<sup>28</sup> between them, which is different from the known intermediate monoclinic structure of Gd<sub>5</sub>Si<sub>2</sub>Ge<sub>2</sub>. Thus, this structure becomes the fourth structure type found in the R<sub>5</sub>X<sub>4</sub> systems.

## Experimental Section

**Syntheses.** The starting materials were pieces of gadolinium (99.99 wt %, Materials Preparation Center, Ames Laboratory), gallium (99.99 wt %, Aldrich), and germanium (99.999 wt %, Aldrich). The alloys with Gd<sub>5</sub>Ga<sub>x</sub>Ge<sub>4-x</sub> ( $x = 0, 0.5, 0.6, 0.7, 0.8, 0.9, 1, 1.5, 2, 2.2, 2.5, 3$ ) stoichiometry and a total mass of up to 3 g were prepared by arc-

melting the element mixtures on a copper hearth in a 116 kPa argon atmosphere. The alloy buttons were remelted six times to ensure homogeneity (weight losses during melting were negligible, <0.1 wt %), and then one-half of each button was wrapped in tantalum foil, sealed in evacuated silica tubes, annealed at 900 °C for 20 h and quenched in cold water.

**X-ray Studies.** The cast and heat-treated samples were characterized by room-temperature X-ray powder diffraction (Enraf Nonius Guinier camera, CuK $\alpha_1$ , Si internal standard). The samples with  $x = 0-2.2$  contained dominant Sm<sub>5</sub>Ge<sub>4</sub>-, Pu<sub>5</sub>Rh<sub>4</sub>- and Gd<sub>5</sub>Si<sub>4</sub>-type phases, and had Gd<sub>5</sub>(Ga,Ge)<sub>3</sub> and Gd(Ga,Ge) impurities with the Mn<sub>5</sub>Si<sub>3</sub>- and CrB-type structures, respectively, which formed upon decomposition of the main phase. Moreover, although annealing improves sample crystallinity, it also increases amounts of the secondary phases. The cast and heat-treated alloys with  $x = 2.5$  and 3 contained Gd<sub>3</sub>(Ga,Ge)<sub>2</sub> (Gd<sub>5</sub>Ga<sub>2</sub>-type) and Gd(Ga,Ge) (CrB-type) phases and were not further investigated. The lattice parameters were derived from the annealed samples by the least-squares method using the CSD program package (Table 1, Figure 1).<sup>29</sup> Pure germanide and mixed gallide-germanides with a low Ga amount ( $x = 0-0.6$ ) adopt a Sm<sub>5</sub>Ge<sub>4</sub>-type structure; the phase with a medium Ga concentration ( $x = 1$ ) belong to a Pu<sub>5</sub>Rh<sub>4</sub>-type structure; the phases with a high Ga concentration ( $x = 1.2-2.2$ ) crystallize in a Gd<sub>5</sub>Si<sub>4</sub>-type structure. Assignment of the Pu<sub>5</sub>Rh<sub>4</sub>-type structure to Gd<sub>5</sub>GaGe<sub>3</sub> is based on the single-crystal refinement of Gd<sub>5</sub>GaGe<sub>3</sub>, which resulted in the interslab T1–T1 bonds of 2.93 Å that are intermediate in length as compared to those in the Sm<sub>5</sub>Ge<sub>4</sub>- and Gd<sub>5</sub>Si<sub>4</sub>-type structures.

Two R<sub>5</sub>X<sub>4</sub>-type phases were observed in the powders with  $x = 0.7, 0.8,$  and  $0.9$ , indicating that the transition from the Sm<sub>5</sub>Ge<sub>4</sub>-type structure to the Pu<sub>5</sub>Rh<sub>4</sub>-type structure is a first-order one. Lattice parameters of the Pu<sub>5</sub>Rh<sub>4</sub>-type phase from these three samples were derived using the least-squares method and are given in Table 1. Although not all lattice parameters for the Pu<sub>5</sub>Rh<sub>4</sub>-type phases with  $x = 0.7, 0.8, 0.9$  and  $1$  are within three standard deviations from one another, it is assumed, however, that the homogeneity range for the Pu<sub>5</sub>Rh<sub>4</sub>-type phase starts at  $x = 1$  (in reality, its lower boundary can be located anywhere in the region of  $0.9 \leq x \leq 1$  and has to be determined experimentally). Only one R<sub>5</sub>X<sub>4</sub>-type phase, except for the Gd<sub>3</sub>(Ga,Ge)<sub>2</sub> and Gd(Ga,Ge) impurities, was observed in the alloys with  $1 \leq x \leq 2.2$ , thus indicating a continuous transition from the Pu<sub>5</sub>Rh<sub>4</sub>-

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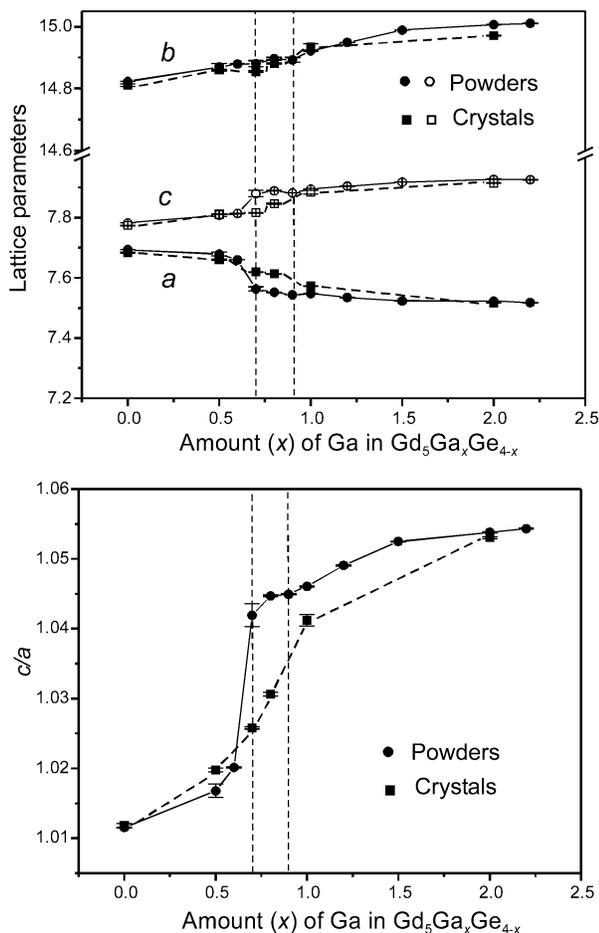
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**Table 2.** Single Crystal Data and Structure Refinement for  $Gd_5Ga_xGe_{4-x}$  ( $Pnma$  space group,  $MoK\alpha$  radiation,  $2\theta$  range =  $4-57^\circ$ ,  $Z = 4$ )

composition <sup>a</sup>	$Gd_5Ge_4$	$Gd_5Ga_{0.5}Ge_{3.5}$	$Gd_5Ga_{0.7}Ge_{3.3}$	$Gd_5GaGe_3$	$Gd_5Ga_2Ge_2$
structure type	$Sm_5Ge_4$	$Sm_5Ge_4$	$Sm_5Ge_4$ 88(2)% + $Pu_5Rh_4$ 12%	$Pu_5Rh_4$	$Gd_5Si_4$
$a$ , Å	7.683(2)	7.660(2)	7.6195(8)	7.572(2)	7.5162(7)
$b$ , Å	14.811(4)	14.860(3)	14.838(2)	14.933(3)	14.971(1)
$c$ , Å	7.774(2)	7.811(2)	7.8162(8)	7.884(2)	7.9149(7)
volume, Å <sup>3</sup>	884.6(4)	889.2(3)	883.7(2)	891.5(3)	890.6(1)
$R [I > 2\sigma(I)]$	$R_1 = 0.0348$ ,	$R_1 = 0.0462$ ,	$R_1 = 0.0651$ ,	$R_1 = 0.0623$ ,	$R_1 = 0.0232$ ,
peak/hole, $e/\text{Å}^3$	2.42 / -2.69	4.79 / -2.78	7.09 / -4.59	3.17 / -2.25	1.99 / -1.32

<sup>a</sup> The compositions presented in the table are those of the initial samples. An EDS quantitative analysis of gallium-containing single crystals gave the following compositions:  $Gd_{5.2(2)}Ga_{0.4(1)}Ge_{3.4(2)}$ ,  $Gd_{5.2(2)}Ga_{0.6(1)}Ge_{3.2(2)}$ ,  $Gd_{5.2(2)}Ga_{1.2(2)}Ge_{2.6(2)}$ , and  $Gd_{5.1(2)}Ga_{1.9(2)}Ge_{2.0(2)}$ , which are within two standard deviations from the compositions of the initial samples, from which they were extracted.



**Figure 1.** Lattice parameters of the  $Gd_5Ga_xGe_{4-x}$  phases as a function of Ga amount. The vertical dashed lines indicate the two-phase region. Powder lattice parameters only of the  $Pu_5Rh_4$ -type structures from the two-phase region are shown. Intermediate values of the lattice parameters for single crystals from the two-phase region indicate an unusual structural behavior.

type structure to the  $Gd_5Si_4$ -type one. The argument that this transformation can be continuous (second-order) is also supported by the Landau theory (not discussed here).<sup>30,31</sup> Therefore, we do not define a transition point between the  $Pu_5Rh_4$ - and  $Gd_5Si_4$ -type structures. The upper boundary of the homogeneity region for the  $Gd_5Si_4$ -type phase extends, at least, to  $x = 2.2$ , which is its last experimentally established existence point.

Single-crystal diffraction techniques were used to confirm powder indexing results and to refine atomic parameters. Crystals were picked from the cast  $Gd_5Ga_xGe_{4-x}$  samples with  $x = 0, 0.5, 0.7, 0.8, 1, 2$  and checked for crystal quality by Laue photographs ( $Cu K\alpha$  radiation). Room-temperature X-ray diffraction data were collected on a Bruker

**Table 3.** Valence Electron Concentrations (per formula unit) and Interlab T1–T1 Bond Lengths for the  $Gd_5Ga_xGe_{4-x}$  Phases

	$Gd_5Ge_4$	$Gd_5Ga_{0.5}Ge_{3.5}$	$Gd_5Ga_{0.7}Ge_{3.3}$	$Gd_5GaGe_3$	$Gd_5Ga_2Ge_2$
st. type	$Sm_5Ge_4$	$Sm_5Ge_4$	$Sm_5Ge_4$	$Pu_5Rh_4$	$Gd_5Si_4$
electrons	31	30.5	30.3	30	29
T1–T1, Å	3.628(2)	3.500(4)	3.461(5)	2.929(7)	2.741(1)

SMART Apex CCD diffractometer with  $MoK\alpha$  radiation and were harvested by taking three sets of 606 frames with  $0.3^\circ$  scans in  $\omega$  and with an exposure time of 20 s per frame. The range of  $2\theta$  extended from  $4^\circ$  to  $57^\circ$ . Intensities were extracted and then corrected for Lorentz and polarization effects through the SAINT program.<sup>32</sup> Empirical absorption corrections were based on modeling a transmission surface by spherical harmonics employing equivalent reflections with  $I/\sigma(I) > 3$  (program SADABS).<sup>32</sup> Structures of the crystals with  $x = 0$  and  $0.5$  were solved by direct methods and refined on  $F^2$  by the full-matrix least-squares method in the  $Sm_5Ge_4$  type, and those of the crystals with  $x = 1$  and  $2$  in the  $Pu_5Rh_4$  and the  $Gd_5Si_4$  types, respectively. Because Ga and Ge atoms cannot be differentiated using X-ray diffraction techniques due to one-electron difference in their electron densities, the same Ge/Ga statistical mixtures consistent with sample stoichiometries were assumed on three sites during the refinement processes. Nearest-neighbor bond distances could not be used to distinguish Ga and Ge atoms either, because similarity in atomic radii makes such analysis fruitless (metallic radii of Ga and Ge are 1.246 and 1.242 Å, respectively).<sup>33</sup> Unit cell dimensions and interlab T–T distances for all investigated crystals are listed in Tables 1, 2, and 3, atomic parameters and isotropic temperature factors only for the crystals with  $x = 0, 0.5, 1$ , and  $2$  are presented in Table 4, interatomic distances for the crystals with  $x = 0$  and  $2$  are shown in Table 5 (additional crystallographic data can be obtained upon request).

The electron density maps for crystals with  $x = 0.7$  and  $0.8$  were very unusual because every peak had a tail (Figure 2). Two additional crystals were picked from the samples with  $x = 0.7$  and  $0.8$  in order to obtain precise atomic positions, but their structure solutions gave similar smeared electron density maps. Presence of a superstructure, which could account for this diffuse electron density, was not supported due to the lack of additional Bragg reflections. Because there was no indication of peak splitting, it was concluded that these crystals are merohedral twins. Treating pear-shape peaks as a superposition of two atoms belonging to two different structure types ( $Sm_5Ge_4$  and  $Pu_5Rh_4$ ) with the same lattice parameters improved the refinement process and led to lower  $R$ -values, e.g., from  $R = 0.116$  to  $0.065$  for the crystal with  $x = 0.7$ . Lattice parameters for the  $Gd_5Ga_{0.7}Ge_{3.3}$  crystal and atomic parameters for the dominant  $Sm_5Ge_4$ -type component are given in Tables 2 and 4.

This twinning is unusual because the two structures have the same unit cell, while preserving individual atomic arrangements. Although indirectly, the intermediate values of the lattice dimensions that fall

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**Table 4.** Atomic Parameters and Isotropic Temperature Factors  $U_{\text{eq}}$  ( $\text{\AA}^3$ ) for Some  $\text{Gd}_5\text{Ga}_x\text{Ge}_{4-x}$  Phases

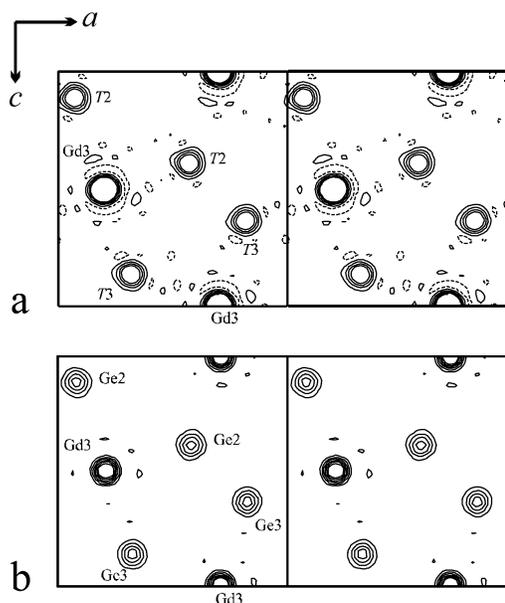
atom		x	y	z	$U_{\text{eq}}$
<b><math>\text{Gd}_5\text{Ge}_4</math> (<math>\text{Sm}_5\text{Ge}_4</math>-type)</b>					
Gd1	8d	-0.02413(7)	0.60009(4)	0.17797(7)	0.0057(2)
Gd2	8d	0.37689(6)	0.11680(4)	0.16143(7)	0.0046(2)
Gd3	4c	0.20985(9)	1/4	0.49915(9)	0.0042(2)
Ge1	8d	0.2178(1)	0.04397(8)	0.4670(1)	0.0054(3)
Ge2	4c	0.0820(2)	1/4	0.1127(2)	0.0053(4)
Ge3	4c	0.3259(2)	1/4	0.8657(2)	0.0050(4)
<b><math>\text{Gd}_5\text{Ga}_{0.5}\text{Ge}_{3.5}</math> (<math>\text{Sm}_5\text{Ge}_4</math>-type)</b>					
Gd1	8d	-0.01572(9)	0.59824(4)	0.18013(9)	0.0117(2)
Gd2	8d	0.36776(9)	0.11817(4)	0.16275(9)	0.0110(2)
Gd3	4c	0.2019(1)	1/4	0.5009(1)	0.0108(2)
T1	8d	0.2097(2)	0.04297(9)	0.4645(2)	0.0125(3)
T2	4c	0.0725(3)	1/4	0.1118(2)	0.0124(4)
T2	4c	0.3174(3)	1/4	0.8657(2)	0.0120(4)
<b><math>\text{Gd}_5\text{Ga}_{0.7}\text{Ge}_{3.3}</math> (<math>\text{Sm}_5\text{Ge}_4</math>-type, dominant component)</b>					
Gd1	8d	-0.0155(1)	0.59773(7)	0.1802(1)	0.0090(2)
Gd2	8d	0.3661(1)	0.11876(7)	0.1638(1)	0.0071(2)
Gd3	4c	0.2006(2)	1/4	0.5019(2)	0.0070(3)
T1	8d	0.2083(3)	0.0424(2)	0.4643(3)	0.0090(4)
T2	4c	0.0712(4)	1/4	0.1118(4)	0.0087(6)
T2	4c	0.3168(4)	1/4	0.8655(4)	0.0074(6)
<b><math>\text{Gd}_5\text{GaGe}_3</math> (<math>\text{Pu}_5\text{Rh}_4</math>-type)</b>					
Gd1	8d	0.0093(2)	0.59592(8)	0.1817(1)	0.0235(4)
Gd2	8d	0.3345(2)	0.12116(8)	0.1705(1)	0.0249(4)
Gd3	4c	0.1694(3)	1/4	0.5085(2)	0.0248(5)
T1	8d	0.1736(4)	0.0409(2)	0.4661(3)	0.0285(7)
T2	4c	0.0379(5)	1/4	0.1095(4)	0.025(1)
T2	4c	0.2883(6)	1/4	0.8679(4)	0.024(1)
<b><math>\text{Gd}_5\text{Ga}_2\text{Ge}_2</math> (<math>\text{Gd}_5\text{Si}_4</math>-type)</b>					
Gd1	8d	0.01709(4)	0.59422(2)	0.18195(3)	0.0112(1)
Gd2	8d	0.32353(3)	0.12204(2)	0.17457(3)	0.0102(1)
Gd3	4c	0.15868(5)	1/4	0.51403(5)	0.0114(1)
T1	8d	0.16010(8)	0.04059(7)	0.46878(7)	0.0120(2)
T2	4c	0.0278(1)	1/4	0.1072(1)	0.0114(2)
T2	4c	0.2742(1)	1/4	0.8711(1)	0.0112(2)

**Table 5.** Interatomic Distances in  $\text{Gd}_5\text{Ga}_2\text{Ge}_2$  and  $\text{Gd}_5\text{Ge}_4$ <sup>a</sup>

atoms	$\text{Gd}_5\text{Ga}_2\text{Ge}_2$		$\text{Gd}_5\text{Ge}_4$	
	distance, $\text{\AA}$	distance, $\text{\AA}$	atoms	distance, $\text{\AA}$
T1–T1( $\times 4$ )	2.741(1)	3.628(2)	T3–Gd1( $\times 8$ )	3.1847(7)
			Gd1( $\times 8$ )	3.2264(7)
T2–T3( $\times 4$ )	2.631(1)	2.683(2)	Gd2( $\times 8$ )	3.0943(7)
			Gd3( $\times 4$ )	2.957(1)
T1–Gd1( $\times 8$ )	3.0624(7)	2.987(1)	Gd3( $\times 4$ )	3.029(1)
Gd1( $\times 8$ )	3.1718(7)	3.123(1)		
Gd1( $\times 8$ )	3.2221(7)	3.244(1)	Gd1–Gd1( $\times 8$ )	3.9095(4)
Gd1( $\times 8$ )	3.5677(7)	3.613(1)	Gd1( $\times 4$ )	4.0406(4)
Gd2( $\times 8$ )	2.9015(7)	2.881(1)	Gd2( $\times 8$ )	4.072(1)
Gd2( $\times 8$ )	2.9318(7)	2.913(1)	Gd2( $\times 8$ )	3.7267(5)
Gd2( $\times 8$ )	3.0289(7)	3.003(1)	Gd2( $\times 8$ )	3.529(1)
Gd3( $\times 8$ )	3.1555(7)	3.062(1)	Gd2( $\times 8$ )	3.8328(4)
			Gd2( $\times 8$ )	3.790(1)
T2–Gd1( $\times 8$ )	3.2849(7)	3.199(1)	Gd2( $\times 8$ )	3.9737(5)
Gd2( $\times 8$ )	2.9823(7)	3.028(1)	Gd2( $\times 8$ )	4.453(1)
Gd2( $\times 8$ )	3.0017(7)	3.075(1)	Gd2( $\times 8$ )	4.1001(5)
Gd3( $\times 4$ )	2.936(1)	2.989(2)	Gd2( $\times 8$ )	3.932(1)
Gd3( $\times 4$ )	3.367(1)	3.161(2)	Gd2( $\times 8$ )	4.2114(5)
			Gd2( $\times 8$ )	4.179(1)
			Gd3( $\times 8$ )	3.6020(4)
			Gd3( $\times 8$ )	3.5627(9)
			Gd3( $\times 8$ )	3.6214(4)
			Gd3( $\times 8$ )	3.6422(9)
			Gd2–Gd2( $\times 4$ )	3.8312(7)
			Gd2( $\times 8$ )	3.946(1)
			Gd3( $\times 8$ )	3.9432(4)
			Gd3( $\times 8$ )	4.081(1)
			Gd3( $\times 8$ )	3.4991(5)
			Gd3( $\times 8$ )	3.4632(9)
			Gd3( $\times 8$ )	3.5247(5)
			Gd3( $\times 8$ )	3.5258(9)

<sup>a</sup> Number of bonds per unit cell is given in parentheses.

between those of the crystals with  $x = 0.5$  and 1 and with the  $\text{Sm}_5\text{Ge}_4$ - or  $\text{Pu}_5\text{Rh}_4$ -type structures, respectively, also indicate an interesting structural behavior. Presence of the two structures in single crystals is likely to result from varying the Ga/Ge ratio within the crystal. Choe et al. observed twinning of two  $\text{Gd}_5\text{Ge}_4$ - and  $\text{Gd}_5\text{Si}_2\text{Ge}_2$ -type phases, which are separated by a two-phase region, in single crystals of  $\text{Gd}_5$ -

**Figure 2.** (a) Electron density contour map at  $y = 1/4$  for  $\text{Gd}_5\text{Ga}_{0.7}\text{Ge}_{3.3}$ . (b) For comparison, electron density contour map at  $y = 1/4$  for  $\text{Gd}_5\text{Ge}_4$ .

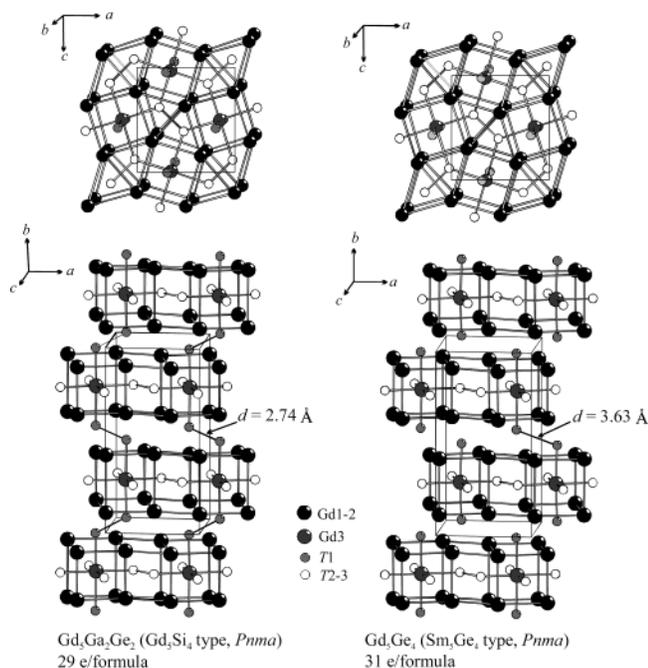
$\text{Si}_{1.5}\text{Ge}_{2.5}$ , and they traced the origin of the twinning to microscopic compositional inhomogeneity within the crystals.<sup>34</sup>

## Results and Discussion

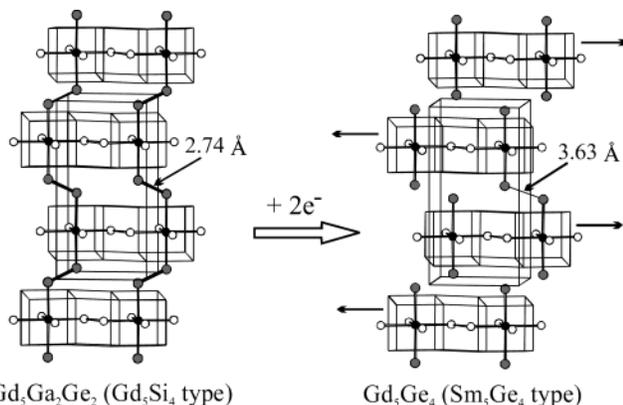
**Structural Changes.** Detailed description of the  $\text{Sm}_5\text{Ge}_4$ -,  $\text{Pu}_5\text{Rh}_4$ -, and  $\text{Gd}_5\text{Si}_4$ -type structures can be found elsewhere.<sup>1,28,34,35</sup> In the  $\text{Gd}_5\text{Ga}_x\text{Ge}_{4-x}$  system, the differentiation between the  $\text{Pu}_5\text{Rh}_4$ - and  $\text{Gd}_5\text{Si}_4$ -type structures is rather technical for the two structures have the same space groups, close lattice constants and similar atomic arrangements. The structural differences are subtle and exhibit themselves as changes in the atomic coordinates (mostly  $x$ ), which lead to shear movement of the  $\infty^2[\text{Gd}_5\text{T}_4]$  slabs (T is a statistical mixture of Ga and Ge atoms on the corresponding sites) and stretching of the T1–T1 interslab bonds. And because there is a continuous transition between the  $\text{Pu}_5\text{Rh}_4$ - and  $\text{Gd}_5\text{Si}_4$ -type structures, the separation of the  $\text{Pu}_5\text{Rh}_4$  type from the  $\text{Gd}_5\text{Si}_4$  type is not clear-cut. To make the relationship between the structures and valence electron concentration more transparent, we treat the  $\text{Pu}_5\text{Rh}_4$ -type structure as the  $\text{Gd}_5\text{Si}_4$ -type one, in which shear movement of the slabs increases the T1–T1 interslab bonds. Thus, we will limit our analysis to the  $\text{Sm}_5\text{Ge}_4$ - and  $\text{Gd}_5\text{Si}_4$ -type structures and will emphasize only the main features of the two structures.

Both crystal structures are built from nearly identical  $3^2434$  nets of Gd atoms (Figure 3). Two such nets are placed over one another along the  $b$  axis to form two-dimensional slabs with Gd3 atoms in pseudo-cubic and T2, T3 atoms in trigonal prismatic voids. Although in the  $\text{Gd}_5\text{Si}_4$ -type (and  $\text{Pu}_5\text{Rh}_4$ -type) phases the slabs are interconnected via covalent-like T1–T1 bonds ( $d_{\text{T1-T1}} = 2.74\text{--}2.93$   $\text{\AA}$ ), in  $\text{Sm}_5\text{Ge}_4$ -type phases all T1–T1 interslab bonds are broken ( $d_{\text{T1-T1}} = 3.46\text{--}3.68$   $\text{\AA}$ ). This bond cleavage is accompanied by shear movement of the slabs along the  $[100]$  direction (Figure 4) and by increase in the corresponding lattice parameter. As judged from the relative atomic arrangements and interatomic distances (see distances

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**Figure 3.** Crystal structures of  $Gd_5Ga_2Ge_2$  and  $Gd_5Ge_4$ , projected along the  $b$  and  $c$  axes. The top projections emphasize the Gd ( $3^2434$ ) nets with the Gd3 in pseudo-cubic and T2–3 in trigonal prismatic voids. In  $Gd_5Ge_4$  the T1–T1 dimers between the slabs are broken.



**Figure 4.**  $Gd_5Ga_2Ge_2$  structure can be transformed into the  $Gd_5Ge_4$  structure through shear movement of the slabs. The interslab T1–T1 dimers break as two neighboring slabs shift in the opposite directions along the  $a$  axis.

for  $Gd_5Ga_2Ge_2$  and  $Gd_5Ge_4$  in Table 5), structural perturbations introduced through the shear movement of the slabs are small inside the slabs but significant between the slabs. A phase transition in the  $Gd_5Ga_xGe_{4-x}$  system can be monitored through the  $c/a$  ratio, as proposed by Choe et al. for the related phases.<sup>34</sup> For the powders the  $c/a$  ratio changes discontinuously with the Ga concentration, thus, indicating a first-order structural transformation (compare the  $c/a$  values at  $x = 0.6$  and  $0.7$  in Figure 1). For the crystals, the increase in  $c/a$  is smoother, with the intermediate  $c/a$  values in the two-phase region, which is due to the unusual structural behavior of the crystals with  $x = 0.7$  and  $0.8$ , as discussed above.

One of the interesting structural features, observed in the  $Gd_5Ga_xGe_{4-x}$  system, is the decrease of T1–T1 interslab bond distances with increase in the Ga amount (Table 3). There are relatively small bond changes within each structure type ( $\sim 0.17$  and  $\sim 0.19$  Å for  $Sm_5Ge_4$ - and  $Gd_5Si_4$ -/ $Pu_5Rh_4$ -type structures, respectively) and a large change ( $\sim 0.53$  Å) upon the phase

transition. Because the Ga and Ge atoms are similar in size but have different numbers of valence electrons, the bond cleavage must result from changes in the electronic structure. It is worth noting that at  $x = 1$  a new intermediate structure is found between the  $Sm_5Ge_4$ - and  $Gd_5Si_4$ -type structures. Although the room-temperature structures in the  $Gd_5Si_xGe_{4-x}$  system change from the  $Gd_5Si_4$  to  $Gd_5Si_2Ge_2$ , and finally to  $Sm_5Ge_4$  type, as  $x$  increases,<sup>23</sup> the structures in the  $Gd_5Ga_xGe_{4-x}$  series evolve from the  $Gd_5Si_4$  to  $Pu_5Rh_4$ , and finally to  $Sm_5Ge_4$  type. Extensive literature search on the  $R_5X_4$  family reveals that  $Ce_{1.22}Sc_3Ge_4$  also has a similar T1–T1 interslab bond distance of 2.95 Å.<sup>36</sup>

**Calculated Electronic Structure of  $Gd_5Ga_2Ge_2$ .** To understand the relationship between the structures and valence electron concentrations, tight-binding linear-muffin-tin-orbital calculations using the atomic sphere approximation (TB-LMTO-ASA)<sup>37</sup> were carried out for the room-temperature structures of  $Gd_5Ge_4$  and  $Gd_5Ga_2Ge_2$ . To satisfy the overlap criteria of the atomic spheres in the TB-LMTO-ASA method, empty spheres were included in the unit cell (44 in  $Gd_5Ge_4$  and 88 in  $Gd_5Ga_2Ge_2$ , employing an automatic sphere generation). The  $4f$  electrons of Gd were treated as core electrons, which is a good approximation due to the fact that both phases are paramagnetic at room temperature (physical properties of the  $Gd_5Ga_xGe_{4-x}$  phases will be reported later).

Two structural models, consistent with the  $Pnma$  symmetry and sample stoichiometry, were considered for  $Gd_5Ga_2Ge_2$ . In the first model, the Ge atoms were placed in the T1 site and Ga atoms in T2 and T3 sites; in the second model, the Ge and Ga atoms were exchanged. Distribution of different atoms over two or more independent sites in a structure is known as a coloring problem.<sup>38</sup> Although electronic and geometric factors usually dictate atomic separation, the entropy contribution to the Gibbs free energy always favors statistical mixture.<sup>39,40</sup> In  $Gd_5Ga_2Ge_2$  size effects can be neglected due to the fact that the atomic radii of Ge and Ga are similar. Therefore, distribution of Ge or Ga atoms over different sites can be qualitatively predicted by comparing total electronic energies of the two models.

In  $Gd_5Ga_2Ge_2$ , all the Ga and Ge atoms form either interslab T1–T1 dimers of 2.74 Å or intraslab T2–T3 dimers of 2.63 Å. According to the Zintl–Klemm electron counting formalism for valence compounds,<sup>41</sup> the dimers are isoelectronic with halogen dimers and carry formal negative charges of either 8 ( $Ga_2$  dimer) or 6 ( $Ge_2$  dimer) because no mixed dimers are present in the two structural models. If Gd atoms are considered as  $Gd^{3+}$ , then the chemical formula can be written as  $(Gd^{3+})_5(Ga_2^{8-})(Ge_2^{6-})(e^-)$ . The remaining electron will occupy T–T  $4p$  antibonding states and also Gd–Gd and Gd–T bonding states. Because Ge is more electronegative than Ga, the  $4p$  antibonding states of  $Ge_2$  dimers are lower in energy and are more populated than those of  $Ga_2$  dimers. Having Ge atoms on the T1 sites, which yield less disperse bands due to a larger

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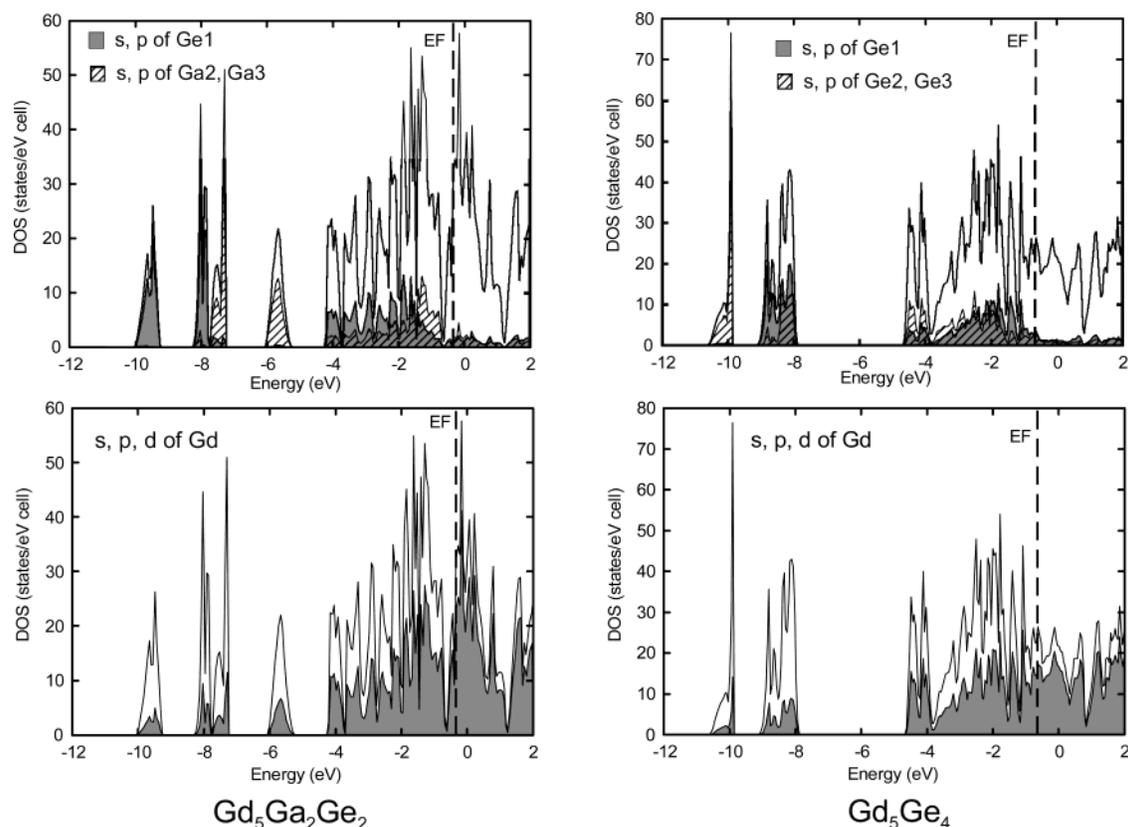
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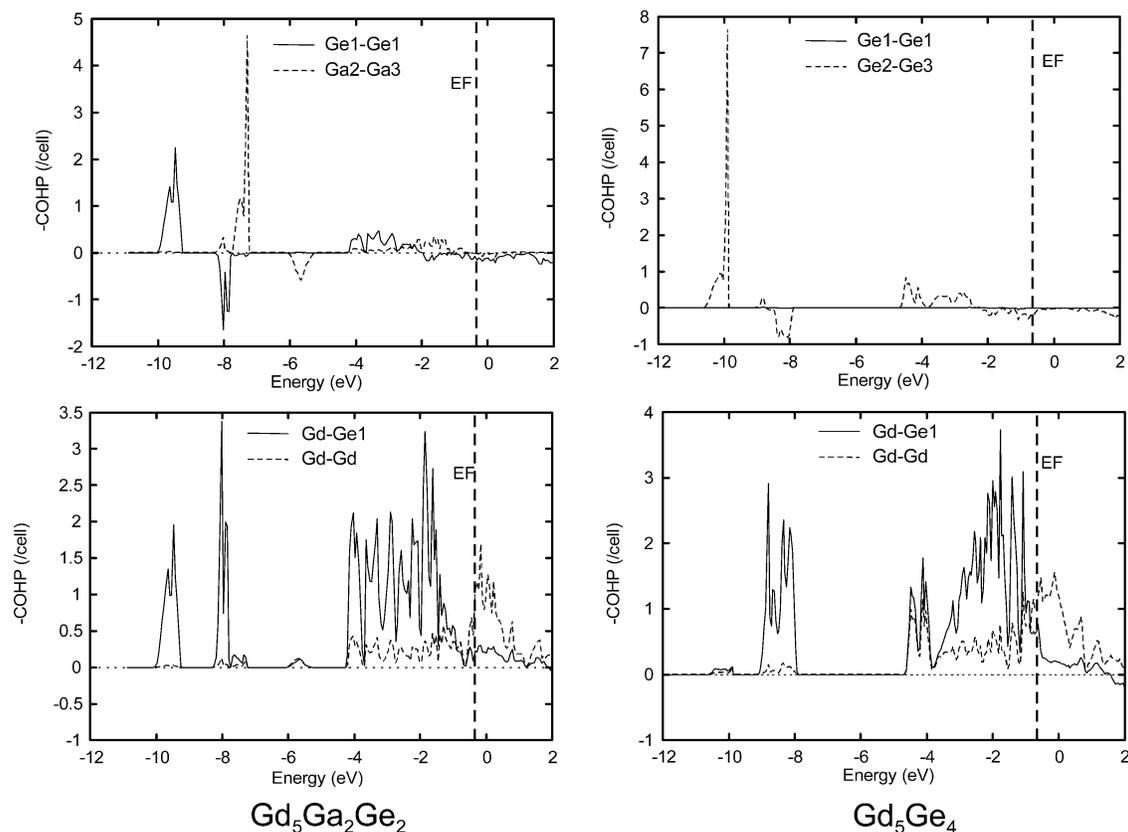
**Figure 5.** Total and projected densities of states (DOS) of  $\text{Gd}_5\text{Ga}_2\text{Ge}_2$  and of room-temperature paramagnetic  $\text{Gd}_5\text{Ge}_4$ . In  $\text{Gd}_5\text{Ga}_2\text{Ge}_2$  Ge atoms are in the T1 site, Ga atoms are in the T2 and T3 sites.

T–T separation, will result in lower electronic energy than placing Ge atoms on the T2 and T3 sites.

This simple reasoning is supported by band structure calculations. The TB-LMTO-ASA method gives lower electronic energy to the first model by 0.35 eV/unit cell, thus indicating a preference for Ge atoms at the T1 site (between the slabs) and Ga atoms at the T2 and T3 sites (inside the slabs). Calculated densities of states (DOS) and crystal orbital Hamilton population (COHP) for the two models are similar and agree well with the qualitative band structure analysis. The DOS and COHP plots for the more stable structural model of  $\text{Gd}_5\text{Ga}_2\text{Ge}_2$  are presented in Figures 5 and 6. Peaks around  $-9.5$  eV,  $-8$  eV and  $-7.5$  eV,  $-6$  eV represent the bonding  $\sigma_s$  and antibonding  $\sigma_s^*$  states of the  $\text{Ge}_2$  and  $\text{Ga}_2$  dimers, respectively, with contributions from the Gd orbitals. The conduction band can be divided into two parts by nearly a pseudo gap at  $-0.6$  eV. The states in the lower part are derived from the  $4p$  bonding states and  $4p$  lone pairs of  $\text{T}_2$  dimers that interact in a bonding manner with the Gd  $6s$  and  $5d$  orbitals, which are also involved in the Gd–Gd bonding. From the integration of the DOS curve, these states and the low-lying  $\sigma_s$  and  $\sigma_s^*$  ones of the  $\text{T}_2$  dimers account for 14 electron pairs per formula unit, which correlates well with the electron counting scheme used above. The states, above  $-0.6$  eV, have the largest contribution from mostly Gd  $5d$  and  $6p$  orbitals, and small contribution from the antibonding  $\sigma_p^*$  states within the  $\text{T}_2$  dimers (intraslab  $\text{Ga}_2$  and interslab  $\text{Ge}_2$ ). Analysis of the bond characters indicates bonding Gd–Gd, Gd–Ge and Gd–Ga, nonbonding intraslab Ga–Ga and antibonding interslab Ge–Ge interactions around the Fermi level (Figure 6).

**Calculated Electronic Structure of  $\text{Gd}_5\text{Ge}_4$ .** Introducing more itinerant electrons into the structure of  $\text{Gd}_5\text{Ga}_2\text{Ge}_2$  will

significantly weaken the interslab Ge–Ge bonds but will have rather a small effect on the intraslab Ga–Ga bonds. Thus, increase in the interslab bond length is expected from electronic considerations and, indeed, is experimentally observed in the  $\text{Gd}_5\text{Ga}_x\text{Ge}_{4-x}$  phases upon substitution of three-valent Ga by four-valent Ge (Table 3). In  $\text{Gd}_5\text{Ge}_4$ , the interslab dimers are considered to be completely broken ( $d_{\text{Ge1-Ge1}} = 3.63$  Å). Treating the Ge monomers to be isoelectronic with noble gas atoms and to carry a formal charge of  $-4$ , we can write the chemical formula of  $\text{Gd}_5\text{Ge}_4$  as  $(\text{Gd}^{3+})_5(\text{Ge}_2^{6-})(\text{Ge}^{4-})_2(1e^-)$ . Presence of the chemically different  $\text{Ge}^{4-}$  monomers with very weak interactions to other  $\text{Ge}^{4-}$  monomers affects the DOS. The two most prominent features of the DOS of  $\text{Gd}_5\text{Ge}_4$  (Figure 5), resulting from the structural changes but not from the Ga/Ge substitution in  $\text{Gd}_5\text{Ga}_2\text{Ge}_2$ , are (i) appearance of an additional DOS peak around  $-9$  eV and (ii) disappearance of the pseudogap, which in  $\text{Gd}_5\text{Ga}_2\text{Ge}_2$  should have been shifted to lower energies. (TB-LMTO-ASA calculations for  $\text{Gd}_5\text{Ge}_4$  in the  $\text{Gd}_5\text{Ga}_2\text{Ge}_2$  structure indicate the band gap shift from  $-0.86$  eV to  $-1.05$  eV. Lower energy of the Ge orbitals, as compared to that of the Ga orbitals, results in the gap shift.) The changes in the DOS are direct consequences of dimer breaking. Since the Ge1–Ge1 interslab interactions are weak ( $d_{\text{Ge1-Ge1}} = 3.63$  Å), the separation between the bonding  $\sigma_s$  and antibonding  $\sigma_s^*$  Ge1–Ge1 states is small. While the antibonding states overlap with the antibonding states of other Ge atoms, the bonding states fall in the energy gap. Small energetic dispersion is also observed for the bonding  $\sigma_p$  and antibonding  $\sigma_p^*$  Ge1–Ge1 states, with the latter moving to lower energies and, thus, eliminating the pseudogap. As in  $\text{Gd}_5\text{Ga}_2\text{Ge}_2$ , the remaining electron in  $\text{Gd}_5\text{Ge}_4$  fills the bonding Gd–Gd, Gd–Ge, and



**Figure 6.** Crystal orbital Hamilton population (COHP) curves for some interactions in  $Gd_5Ga_2Ge_2$  and in room-temperature paramagnetic  $Gd_5Ge_4$ . In  $Gd_5Ga_2Ge_2$  Ge atoms are in the T1 site, Ga atoms are in the T2 and T3 sites. Interactions in the upper part are bonding, in the lower part antibonding.

antibonding intraslab Ge2–Ge3  $sp^*$  states. But, since the energy of Ge orbitals is lower than that of Ga orbitals, the antibonding  $sp^*$  states of the dimers are now populated in  $Gd_5Ge_4$ . This argument is also valid for the Ga-containing phases, since there is always a mixture of Ga and Ge atoms on T2 and T3 sites. As a result, an increase in the T2–T3 bond distances is expected and, indeed, is experimentally observed on going from  $Gd_5Ga_2Ge_2$  to  $Gd_5Ge_4$  in the  $Gd_5Ga_xGe_{4-x}$  system (see Table 5).

Reduction in the Ge1–Ge1 orbital overlap upon dimer cleavage leads to strengthening of Gd–Ge1 bonds in  $Gd_5Ge_4$ . Optimization of Gd–Ge1 interactions is intuitively expected from chemical considerations, because the Ge1 electrons, freed from bonding in the  $Ge_2$  dimers, are donated to the Gd–Ge1 interactions. The COHP calculations support this argument. Appearance of additional states in the Gd–Ge1 bonding region from  $-1$  eV to  $-0.5$  eV is a direct consequence of the Ge1–Ge1 bond cleavage (Figure 6). Increase in the Gd–Ge1 bonding correlates well with the changes in the Gd–Ge1 interatomic distances: while in  $Gd_5Ga_2Ge_2$ , the average Gd–T1 distance is  $3.1302(7)$  Å, in  $Gd_5Ge_4$  the average Gd–Ge1 distance is  $3.103(1)$  Å. Thus, there is an energetic tradeoff in interactions upon transforming the  $Gd_5Ga_2Ge_2$  structure into the  $Gd_5Ge_4$  one: while the interslab T1–T1 interactions became weaker, the Gd–T1 bonds became stronger (calculated  $-ICOHP$  values for Gd–Ge1 interactions are  $5.39$  and  $6.46$  eV/cell for the  $Gd_5Ga_2Ge_2$  and  $Gd_5Ge_4$ , structures respectively).

Interestingly, the T1–T1 bond does not stretch gradually in the  $Gd_5Ga_xGe_{4-x}$  system. Increase in the electron concentration by  $1 e^-$ /formula unit results in a modest dimer stretching by  $\sim 0.19$  Å on going from  $Gd_5Ga_2Ge_2$  to  $Gd_5GaGe_3$ , but introduc-

ing an extra  $0.3$  electron into  $Gd_5GaGe_3$  leads to complete dimer cleavage ( $d_{T1-T1} = 3.46$  Å) in  $Gd_5Ga_{0.7}Ge_{3.3}$  and to the first-order phase transition. Further increase in the electron concentration is followed again by a small stretching ( $\sim 0.17$  Å) of the T1–T1 bonds. At present, it is not fully understood why there is a sudden change in the T1–T1 interactions instead of gradual bond stretching. It can be assumed that a structure with intermediate T1–T1 distances is unstable with respect to the  $Pu_5Rh_4$ - and  $Sm_5Ge_4$ -type structures. However, the dependence of the interslab T1–T1 distances and structures themselves on the number of valence electron electrons is quite obvious in the  $Gd_5Ga_xGe_{4-x}$  system: the phases with low valence electron concentrations adopt the  $Gd_5Si_4$ -type structure with T1–T1 dimers between the slabs, the phase with medium valence electron concentrations belong to the  $Pu_5Rh_4$ -type structure with intermediate interslab T1–T1 distances, and the phases with high valence electron concentrations have the  $Sm_5Ge_4$ -type structure with broken interslab T1–T1 dimers. This argument can be extended to other  $R_5X_4$  phases and may be utilized in predicting and, subsequently, obtaining new phases. In our view, introducing extra electrons into the silicon rich  $Gd_5Si_xGe_{4-x}$  compounds with the  $Gd_5Si_4$ -type structure is likely to yield phases with broken interslab bonds.

## Conclusions

Structural transformations in the  $Gd_5Ga_xGe_{4-x}$  system reveal an intimate relationship between the crystal structure and its valence electron concentration. Increase in electron concentration through substitution of four-valent germanium for three-valent gallium results in larger population of antibonding T1–T1 states

and, consequently, in stretching and breaking the T1–T1 interslab dimers. Dimer cleavage is accompanied by the shear movement of the slabs.

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**Supporting Information Available:** Crystallographic data relevant to the crystal structure refinement of the  $\text{Gd}_5\text{Ga}_x\text{Ge}_{4-x}$  single crystals. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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